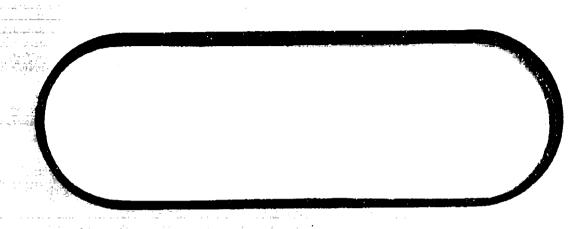
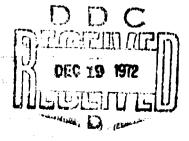
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USAAVLABS Technical Report
November, 1972

TEST RESULTS REPORT AND FINAL TECHNOLOGY DEVELOPMENT REPORT, HLH/ATC TRANSMISSION HOUSING MATERIAL EVALUATION

FINAL REPORT

Ву

L. Kozella/D. West

Prepared by

The Boeing Company Vertol Division

for

U. S. ARMY AVIATION MATERIEL LABORATORIES FORT EUSTIS, VIRGINIA

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SUMMARY

The majority of the temsile, fatigue, and crack propagation test results were obtained from ZE41A magnesium sand castings which were subjected to a one-stage precipitation heat treatment cycle of 480°F for 24 hours. This treatment was developed by experiment at the vendor's facility and involved the testing of 108 separately cast tensile test coupons after exposure to four different thermal cycles including the two-stage treatment recommended by Magnesium Elektron of England who introduced and developed this alloy. Mechanical test results of thin and thick material from all castings of ZE41A-T5 given the one-stage treatment met or exceeded AMS 4439 requirements. Similar tests on ZE41A-T5 casting which was given a two-stage thermal treatment exhibited average yield strength values below AMS requirements.

Tensile tests of all the nonwelded ZE41A-T5 specimens, even those removed from 3-inch-thick sections, indicated strengths and elongations in excess of AMS 4439 requirements (Figure 2). All the welded specimens, with the exception of those removed from the thickest section, exhibited mechanical properties exceeding the requirements for nonwelded material. Maximum thickness of the HLH transmission housings is anticipated to be from 1.5 to 2.0 inches.

Bending fatigue tests of the "as-cast" surface of nonwelded ZE4LA-T5 specimens indicated that those from the thin section of the casting had an average fatigue strength approximately 20 percent higher than those from the thick section of the casting (Figure 5). This difference in fatigue performance is attributed to the surface condition and microstructural factors. This fatigue testing conducted on the ZE4LA-T5 material indicates that the current fatigue design allowable stress currently used for AZ91C-T6 transmission housings is applicable and is conservative for ZE4LA-T5.

The thick sections revealed a problem of surface finish and microstructure associated with the geometry of the casting which is similar to an ingot of magnesium 16 inches x 14 inches x 30 inches. Chill bars used for cooling this mass of metal could only be located on the drag surface (bottom) since the cope half of the mold contained several risers. In actual production of transmission housings, this particular problem would not occur. Due to the geometry of thick sections, such as transmission mounting pad areas, it is possible to place chills on three or four sides of the pad. This permits optimum cooling rates and should limit the degree of segregation in the microstructure. The optimum test casting for this type of evaluation should have an irregular profile with 3-inch-wide bars having variable heights. This shape would permit the foundry to chill the casting in a manner

more representative of good casting technology.

Welding proved detrimental to the fatigue strength of ZE41A-T5 magnesium. Welded specimens from both thin and thick sections of the casting showed a 10 to 20 percent lower fatigue strength than nonwelded specimens from corresponding sections. It should be noted, however, that currently weld repair is permitted only in the low stress, noncritical areas of a helicopter transmission housing. Under this condition, weld repair offers an effective economic means of salvaging many castings without compromising structural integrity. The testing described herein indicates that this approach could also be utilized with HLH transmission housings of ZE41A-T5.

Fatigue crack propagation testing of ZE41A-T5 material conducted in air at 10 Hz indicated essentially the same crack growth rate characteristics for welded and unwelded specimens utilizing material from thick or thin sections of the casting (Ligure 20). Tests conducted to compare the fatigue crack growth characteristics of ZE41A-T5 and AZ91C-T6 indicated that the AZ91C-T6 material had a crack growth rate which, on the average, was two to three times slower than that of the ZE41A-T5 material. While this difference is primarily attrihated to microstructural differences and is considered signifigant, there is a question as to how well the AZ91C-T6 material used in this program represented yield strength and elongation properties of actual transmission housing castings. The low yield strength and high elongation of the material are not considered representative of typical properties determined from tests on material removed from actual housings.

Since minimum wall thickness and stiffness requirements, rather than strength requirements, control the design of a typical helicopter transmission housing, small differences in strength properties from alloy to alloy are generally not agnificant. The previous successful structural performance of AZ91C-T6 housings is an indication that ZE41A-T5 housings having similar mechanical properties will also perform satisfactorily. In addition, ZE41A's superior castability, resulting in less scrappage, will provide a more cost-effective design.

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INTRODUCTION

Magnesium castings have been extensively used in the helicopter industry as a housing material for transmissions. The ZE41A-T5 magnesium sand casting alloy is reported by aircraft users in both the United Kingdom and Europe to be cost-effective due to virtual elimination of the microporosity problem which severely affected component deliveries of AZ91C-T6 alloy castings. Magnesium alloys containing rare earths, such as cerium in ZE41A-T5, are rated above the aluminum zinc alloys for general castability. Due to improved castability and minimal microporosity, mechanical properties should have improved uniformity. Since rejections of ZE41A-T5 castings will be lower, overall costs should compare with AZ91C-T6. In addition, the process of impregnation, which is applied to all AZ91C-T6 castings, will not be required for ZE41A-T5 due to the lack of microporosity in this alloy.

With the proposed usage of ZE41A magnesium for the HLH transmission housings, it became necessary to generate reliable static and fatigue strength properties using test coupons prior to fabrication and testing of full-scale transmission components. The proposed housings, fabricated from large size magnesium castings, have various thicknesses. To evaluate thickness effect on material properties such as fatigue strength, specimens were fabricated from both thick (3.0 inches) and thin (0.3 inch) areas of the test castings.

Weldments are not expected or designed into the transmission housings; however, repairs due to sand holes, chisel gouges, gas pockets and other foundry-produced defects invariably occur in production castings. Therefore, to evaluate the effect of welds on fatigue strength, tensile strength, and fatigue crack propagation rates, welds were made in the test castings to simulate such repairs. Figure 1 shows each type of specimen configuration after welding.

Fatigue crack propagation tests of welded and nonwelded magnesium specimens, fabricated from both AZ91C and ZE41A, were conducted to compare the crack propagation rates and to generate data for both materials. This data can be applied to various geometric models or components to determine fatigue crack growth response.

The tests were carried out over the time period from December 1971 to June 1972.

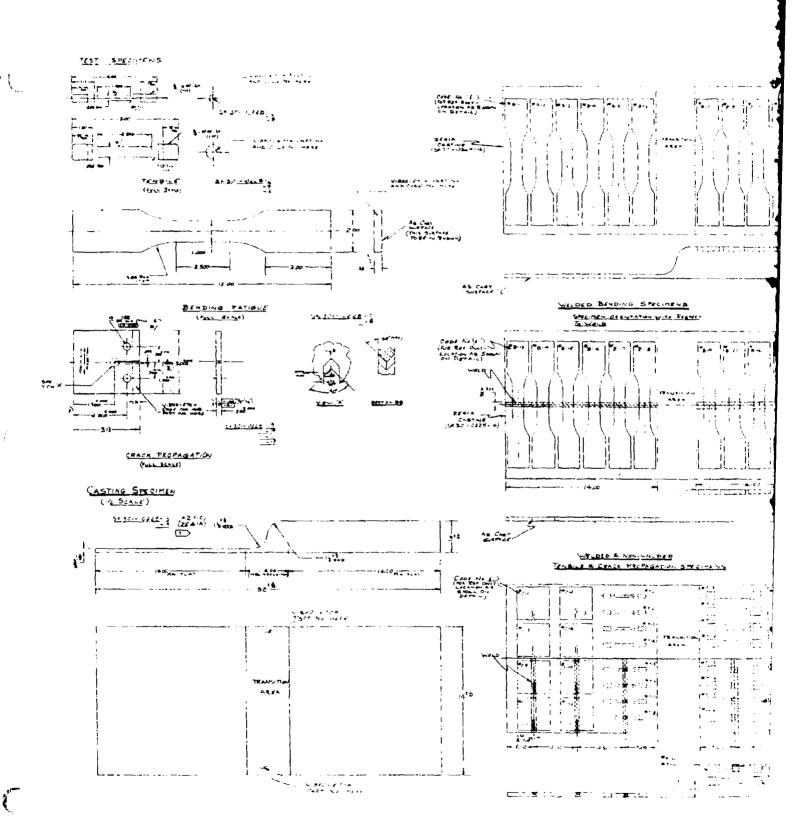
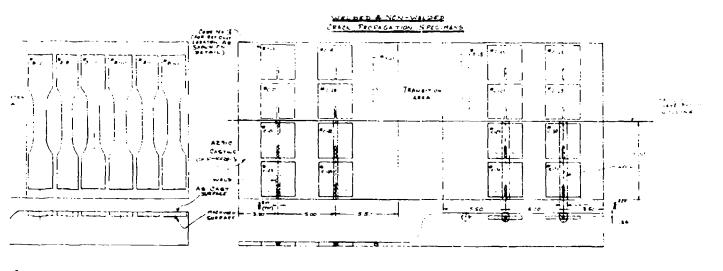
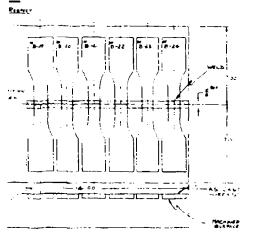


Figure 1. SK301-10228, Housing Materi





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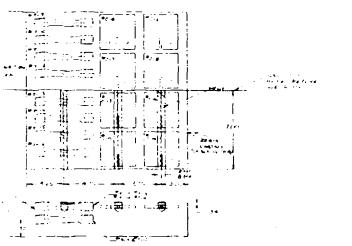


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TEST PROGRAM

PROGRAM PLAN

All test coupons were designed to be removed from four-step (slab) castings per SK301-10228 (Figure 1). Three -14 parts furnished all the fatigue, tension, and crack propagation specimens of the magnesium alloy ZE41A-T5. A fourth casting designated -13 was cast of AZ91C-T6 magnesium for crack propagation and tension specimens. The weldments for the ZE41A tensile, fatigue, and crack propagation specimens were simulated repair TIG (tungsten, arc, inert gas) welds made per MIL-W-18326 using rods of ZE41 per AMS 4439. Welds were ground flush with the casting surface, x-ray inspected, and heat treated to the T5 condition to simulate weld repairs of a transmission or production housing. Welds in the A291C specimens were made using an AZ92 welding rod. After welding, the specimens were heat treated to the T6 condition per MIL-H- ! 6857. Test specimen configuration, weld geometry and related specifications and special nondestructive test requirements were identified on the drawing. Position of the specimens relative to the mass of the casting was established together with the specimen identity.

Testing per drawing included a total of 74 specimens divided as follows:

ALLOY	SPECIFICATION	TENSION	FATIGUE	CRACK PROPAGATION
ZE41	AMS 4439	16	24	16
AZ91	QQ-M-56	2		16

In all cases except for AZ91C-T6 tensile coupons, specimens were equally divided between thick and thin areas as well as welded and nonwelded configurations per the drawing.

MATERIAL

Five castings were utilized in the coupon test program. Three ZE41 castings and one AZ91 casting were furnished by Vendor A. A fifth ZE41 test casting was furnished in the welded and nonwelded condition by Vendor B. A summary of chemical analysis and heat treat batch numbers of the four Vendor A castings are included in Table I. The AZ91 casting was heat treated per MIL-H-6857. Since this specification does not presently cover ZE41 material, tensile tests on 108 separately cast bars were conducted at Vendor A facility prior to heat treating the test castings. Based on these results, the most consistent average and individual mechanical properties were obtained by using a thermal cycle of 480° F

for 24 hours. Vendor A ZE41 castings all received this treatment. The ZE41A-T5 casting furnished by Vendor B was heat treated for 2 hours at 6250F followed by aging for 16 hours at 3450F.

TAI	BLE I.	CHEMICA	L ANALYSI	ES FRO	OM VENDO	OR CERTIFIC	CATION
				Comp	osition	- Percent	
	Melt	Heat Treat Batch			Total Rare Earth		
Alloy			Aluminum	Zinc		Zirconium	Magnesium
3041A	J2011M	M008		4.3	1.4	.75	Balance
ZE 4 lA	J2211M	M008		4.2	1.1	.71	Balance
ZF4 1 A	L3011M	1122-5	~	4.3	1.2	.74	B alance
AZ 31C	L2917R	1102-6	8.6	.80			Balance
1	139 Requ 2541A)	uirement		3.5	1.75	1.0	Bala ice
	56 Requ. (AZ910)	irement	8.1	1.0			Balance

All applicable specifications and technical requirements for the test castings are conveyed by general notes on the engineering drawing (Figure 1). Flag note number 7 speciied that inspection should be accomplished per D8-1059. This ancluded acid etching of all cast surfaces prior to perecrant inspection. Inadvertently, this step which aids identification of defects was not done. Penetrant and radiographic inspections were subsequently completed and all test castings were certified to be acceptable. Radiographs of the 3.0 inchthick area did not show the aurface pits, oxides or secregation characteristics due to the low sensitivity of 2 persent or 0.060 inch penetration in the 3-inch section. These d.screpancies were not observed until the failed fatigue specimens were examined fractographically and by x-ray. Present acceptance practice of inspection with the AZ91C-T6 alloy is hased entirely on the radiographic quality. When a surface discrepancy is evident by fluorescent penetrant inspection, radiographs are used to judge whether to accept or reject the casting.

TENSILE TEST

Test Specimen Configuration and Preparation

Specimens were tested in the following conditions:

- Unwelded from thick, 3.0 inches, and thin, 0.3 inch, sections of a ZE41A casting.
- Welded from thick, 3.0 inches, and thin, 0.3 inch, sections of a ZE41A casting.

Sixteen specimens, four for each condition and thickness, were tested.

Thin specimens were machined to the R3 configuration and thick specimens to the R1 configuration of Federal Test Method Standard Number 151, Method Number 211. End portions of the 3 inch and 5-1/2 inch lengths were threaded for 5/8 inch and 1 inch lengths with 3/8-16 NC and 3/4-10 NC threads, respectively (SK301-10228-1 through -6).

Test Procedure and Test Setup

Tests were performed on a Baldwin Lima Hamilton machine with a 60,000-pound capacity. Grip end attachments were threaded, and axial alignment was achieved by a self-aligning feature in the top and bottom heads. The indicated loads were maintained within 1 percent of the true applied loads. A strain rate of 0.0005 inch/inch/minute was utilized.

Test Results

Tensile test results on 22 specimens of ZE41A-T5 and AZ91C-T6 material are shown in Table II. Average tensile properties for ZE41A-T5 magnesium are shown graphically in Figure 2. All eight coupons from the 0.3-inch-thick area of the casting met minimum and average requirements specified in AMS 4439. Four of these were notched and welded to simulate casting repair procedures. Four of the thick coupons representing the basic magnesium casting at locations both near the surface and at midthickness (No. TK9 and TK10) also met average and individual AMS requirements. Thick welded coupons met yield strength requirements applied to unwelded coupons. However, all tensile strengths and three elongation values for these specimens were below unwelded specification limits. The average tensile strength for welded specimens was 15 percent below specification limits, and clongation values for welded specimens were 30 percent below the requirement of 2.5 percent for unwelded material. It is apparent, based on this test data from one casting, that the magnesium alloy of zinc and rare earth elements has uniform mechanical properties

TABLE II. TENSILE TEST RESULTS FOR ZE41A-T5 AND AZ91C-T6 MAGNESIUM

<u> </u>				·						
Material	Section of Casting	Specimen Descrip- tion			Area (in ²)	Load (1b)	UTS (ksi)	Load (1b)	YTS (ksi)	% E- Long. in, in
			1		.050	1756	35.1	1075	21.5	4.0
		Non-	2		.050	1530	30.6	1060	21.2	4.5
		w ded	3	.252	.050	1502	30.0	1040	20.8	5.0
	— 1.		4	.252	.050	1446	28.9	1030	20,6	3.5
	Thin		5	.251	.0495	1384	28.0	1030	20.8	3.0
[Welded	6	.251	.0495	1332	26.9	1060	21.4	2.0
			7	.252	.050	1590	31.8	1120	22.4	5.0
0.0412			8	.252	.050	1646	32.9	1080	21.6	2.5
ZE41A-T5			9	.505	.2003	5600	28.0	4080	20.4	2.5
(1)		Non-	10	.505	2003	6174	30.6	3900	19.5	5.5
		welded	11	.506	.201	5940	29.6	3950	19.7	4.5
į			12	.506	.201	5800	28,9	3850	19.2	4.5
1	Thick		13	.506	.201_	5090	25.3	3900	19.4	1.0
{		W-1-2-3	14	.506	.201	476C	23.7	4100	20.4	1.0
<u> </u>		Welded	15	.505	.2003	5162	25.8	4160	20.7	2.0
L			16	.506	.201	3767	18.7	3767	18.7	1.0
	Thin		17	.251	.0495	1668	33.7	1090	22.0	*
ZE41A-T5	111.211	Non-	18	.251	.0495	1648	33.3	1050	21.2	*
(2)	Thick	welded	19	.506	.201	5970	29.7	3618	18.0	*
		L	20	.505	.2003	5970	29.8	3605	18.0	*
AZO' - T6	Thin	Non-	21	.252	.050	2025	40.5	740	14.8	12.0
^•	Thick	weldea	22	. 506	.201	7517	37.4	2650	13.2	13.0

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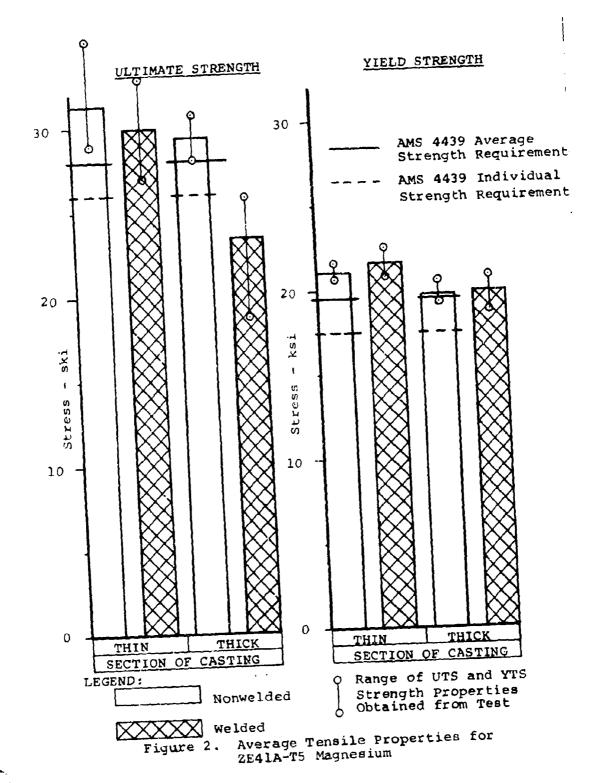
	AMS 4	4439 Requirements	UTS	YTS	% Elong'n
ZE41A-T5		Average	28.0	19.5	2.5
		Individual	26.0	17.5	2.0

	QQ-M-56 Min. F	Requirements	UTS	YTS	% Elong'r
AZ 91C -T6	Average		25.5	14.5	0.75
10	Individual		17.0	12.0	

(1) All material heat treated 480°F for 24 hours

Supplied by Vendor B - Casting was heat treated in two stages; 625°F--2 hours followed by 16 hours at 345°F

(3) Reference Drawing SK301-10228, Figure 1



which meet or exceed AMS requirements in both thick and thin areas. This indicates that there should be no restrictions on the part of foundries meeting properties in sections to 3-inches. Most sources, however, will take exception to AMS strength requirements in sections greater than 12-inches. Guarantee will be made to meet 80 percent of the published values.

Tensile tests in addition to the proposed work plan included two AZ91C-T6 specimens machined from a Vendor A casting and four ZE41A-T5 specimens machined from a Vendor B slab type casting. These results are also tabulated in Table II.

Both of the AZ91C-T6 test specimens met QQ-M-56 minimum requirements. Yield strength values for ZE41A-T5 specimens removed from thick areas meet AMS requirements for individual bars but do not meet average value requirements.

FATIGUE TEST

The purposes of the fatigue testing of the ZE41A magneshum were threefold:

- 1. Determine if a difference in mechanical properties and/or fatigue strength exists between thin and thick sections of magnesium castings.
- 2. Determine if weld repair of magnesium is detrimental to fatigue strength.
- Compare the fatigue strength of ZE41A magnesium with the fatigue strength of AZ91C.

Test Specimen Configuration and Preparation

Twenty-four ZE41A bending specimens were fabricated from large size castings as shown in Figure 1. These specimens, incorporating a "dog-bone" shape, were fabricated as one unwelded specimen cut from both thin and thick sections of the casting and two welded specimens cut from both thin and thick sections of the casting. Four groups, each consisting of six specimens and each representing one of the above conditions, were tested.

After machining the specimens to the dimensions specified by drawing SK301-10028-7 and -8, the edges were broken to approximately 0.010-0.030 inch radius. For all specimens except where noted, the surface and weld areas of the test specimens were tested with the "as-cast" and "as-cast plus welded" surfaces undisturbed. The weld beads and adjacent material on specimens B-14, B-16, B-17, and B-18 were surface dressed.

All nonwelded fatigue test specimens were of the configuration shown in Figure 1. This specimen configuration was to

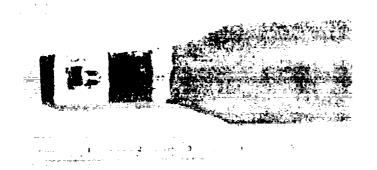
be used for all fatigue specimens, i.e., welded and nonwelded. Testing of three welded specimens (B-22, B-23, B-24) of this configuration resulted in two failures which did not pass through the weld. Examination of specimens B-22 and B-23 indicated considerable variation in the specimen thickness over the length of the test section. This thickness variation was due to irregularities associated with the "as-cast" surface (the other surface was machined). These particular welded specimens had the thickest portion of the test section at the Since the test setup is designed to produce the weld area same bending moment over the entire length of the test section, the thicker sections (the width is constant) have a greater section modulus and, therefore, experience a reduced bending stress. This situation was not conducive to meeting one objective of the test which was to evaluate the relative fatigue strength of the welded and unwelded material.

In order to expose both the parent and weld material to the same stress and thereby determine their relative performance, the specimen configuration was modified. In order to achieve the same stress along the length of the test section, the sides of specimens B-13, B-14, B-16, B-17, and B-18 were machined to provide a relatively constant section modulus. Specimen B-13 was inadvertently machined to the configuration shown in Figure 3. This configuration was not considered satisfactory because of the secondary radius; therefore, it was rejected because of the introduction of an increased stress concentration factor. This configuration was then reworked into a specimen with a gradual or constant radius as displayed in Figure 3. These specimens with this gradual radius (B-14, B-16, B-17) had all surfaces mistakenly machined.

Test Procedure and Test Setup

The fatigue tests of the 24 ZE41A specimens were performed on a Sonntag SF-1 U Fatigue Machine with a standard bending fixture utilizing four-point loading as displayed by Figure 4. All tests were performed at 30 cycles per second loading rate, and at a stress ratio (ratio of minimum load/maximum load) equal to zero. Tests were conducted in air at room temperature.

All specimens were strain gaged at their centerlines and mounted in a static four-point bending fixture for calibration. Known loads, as measured by a load cell, were applied at the constant bending section of the specimen to establish a bending strain output versus load input. Although the calibration data was not used in determining the applied stresses, it did serve a purpose. By using the calibration data, stresses calculated by multiplying modulus of elasticity times strain output (E E) can be checked for accuracy and



SPECIMEN B-1 (Parallel Sides)

SPECIMEN B-13 (Radius at Test Section)





SPECIMEN B-14
(Sides Contoured for Constant Section Modulus)

NOTE: Photographs show view of bottom surface of specimens after fatigue testing.

Figure 3. Fatigue Test Specimen Configurations

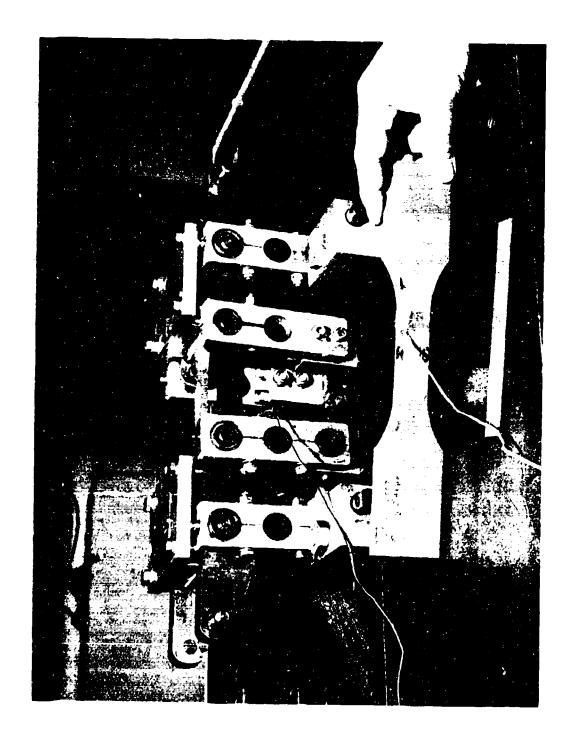


Figure 4. Fatigue Test Bending Fixture

errors by using the equation G = Mc/I.

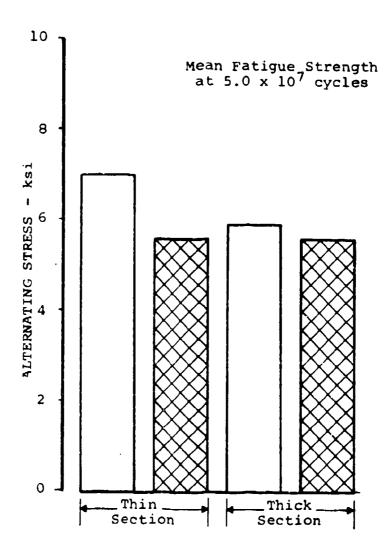
The strain gage for each specimen was bonded to the compression loaded surface. The "as-cast" surface, for all specimens except B-14, B-16, and B-17 which had all surfaces machined, was subjected to tensile stresses created by the applied bending moment. The specimens were stressed to a predetermined level by applying loads and monitoring the strain output. Knowing the modulus of elasticity to be 6.5 x 10⁶ psi for ZE41A, the stress was calculated as E x & . This is the stress shown on the S-N (stress-number of cycles) curves (Figures 6 through 9).

Test Results

The results of the fatigue tests are summarized graphically in Figure 5. The detailed test data are presented in Tables III and IV, and displayed by the S-N curves on Figures 6 through 9.

Comparison of nonweld fatigue data obtained from specimens fabricated from the thin section of the casting (Figure 6) with data obtained from specimens fabricated from the thick section of the casting (Figure 7) shows a significant difference in fatigue strength. The estimated average fatigue strength is +7000 psi at 5 x 107 cycles (at a stress ratio of zero) for specimens from the thin section, and +5900 psi for specimens from the thick section. The data shown in Figures 6 and 7 were obtained from unwelded specimens and from welded specimens which did not fail in the weld or heat-affected zone. A probable explanation for the difference in fatigue performance is the relatively poorer surface quality of the thick section of casting. The porosity and segregation assoclated with the thick section proved to be the origin of failure in many specimens and is believed to have caused the fatigue strength to be lower than that of the thin section with a surface relatively free of porosity.

Side-by-side photographs illustrating the microscopic fracture characteristics plus the origin location and microstructural details of the origin sites are displayed in Figures 10 through 13. Dimensional aspects of fatigue origins on all specimens relative to the spanwise length and section thickness are tabulated in Table V. For specimens representing the thin section of casting, subsurface origins were apparent on specimens B-1, B-2, B-4, and B-5. Of these, B-1, B-4, and B-5 were directly related to gross oxide skins of zirconium or a combination of zirconium and cerium. The microstructure of specimen B-2 was unique since the origin area displayed both entrapped oxides and enlarged grains.



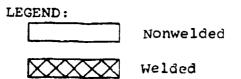


Figure 5. Mean Fatigue Strength for ZE41A-T5 at a Stress Ratio of Zero.

TABLE III. HIH FATIGUE TEST DATA FOR ZE41A MAGNESIUM

Number Strain Gage Applied Bending Number of East (psi) Number Remarks Number Steady acting Alter failure Failure Remarks L177 1277 8200 8300 0.268 Radius B-1 1378 1368 1360 0.268 Radius B-2 1308 1308 8500 0.268 Radius B-3 1462 1462 9500 0.228 Ranout B-3 1462 1462 9500 0.228 Ranout B-4 1262 8200 900 0.228 Ranout B-4 1262 8200 900 11.137 Ranout B-5 1260 8200 900 10.00 10.00	e O	Casting Se	Section: Th	Thin	Stress	Ratio: R=	R=O Test	Frequency: 30 cps
Number Steady Stess Cycles C			ند	Gage	Applied	Bend	Number	
Number Steady nating X10 ⁶ B-1 1277 1277 8300 8500 0.268 Failure at Radius B-3 1155 1308 8500 0.268 Failure in Test Se B-3 1155 1462 9500 9500 0.222 Failure in Test Se B-4 170 770 9500 9500 0.222 Failure in Test Se B-4 170 770 9500 9500 0.222 Failure in Test Se B-4 1262 1385 9500 9500 0.222 Failure in Test Se B-4 1385 1385 9500 9000 20.774 Failure in Test Se B-4 1385 1385 9500 9000 32.857 Runout B-5 1383 1383 8800 8000 9700			Readi	ng ch/Inch	orress Q=		Cycles to	
B-1 Steady nating Steady nating Steady nating Steady nating XIV B-1 1277 1277 1277 8300 8300 0.268 Failure in Test Se B-2 1308 1308 1500 7500 10.316 Runout B-3 1462 1462 9500 9500 0.222 Failure in Test Se B-4 770 770 8200 900 900 10.40 Runout B-4 1262 1262 8200 900 900 11.137 Runout B-4 1262 1262 8200 9200 900 11.137 Runout B-5 1262 8200 8200 13.86 Runout 1651 1651 17.85 18.86 B-5 1362 8200 8200 10.00 10.00 Runout 1651 1651 1651 1651 1651 1651 1651 1651 1650 1650	Specimen	Number	21212	Alter-		Alter-	Failure	Remarks
B-1 1277 1277 1277 1277 1277 1277 1277 1277 1279 8300 0.256 Failure in Test Se B-2 1308 1308 1300 6500 0.222 Failure in Test Se B-3 1462 1462 750 9500 0.222 Failure in Test Se B-4 1262 1262 8200 900 0.227 Failure in Test Se B-4 1262 1262 820 900 20.73 Runout B-5 1230 1200 900 20.74 Failure in Test Se B-5 1262 820 900 32.857 Runout B-5 1262 820 900 10.00 10.00 B-5 1262 820 900 10.00 10.00 B-5 1353 1353 890 900 10.00 10.00 B-5 1362 1360 850 10.00 10.00 10.00 B-13* <td>חבירולידיייי</td> <td></td> <td>Steady</td> <td>nating</td> <td>Steady</td> <td>nating</td> <td>×10</td> <td>ibed to over</td>	חבירולידיייי		Steady	nating	Steady	nating	×10	ibed to over
B-2 1308 1308 8500 0.258 Fallure in Test Segon B-3 1155 7500 7500 10.316 Runout B-3 1462 1462 1462 9500 9500 10.640 Runout B-4 770 770 8200 8200 31.137 Runout B-4 1262 1262 8200 8000 8000 20.774 Failure in Test Segon B-5 1262 1262 8200 9000 20.774 Failure in Test Segon B-5 1262 8200 9000 9000 32.87 Runout B-5 1385 1385 9000 9000 30.33 Failure at Radius B-6 1385 1303 1308 8500 9000 9.700 6.439 Failure at Radius B-13* 1303 1308 8500 6500 0.303 Failure at Radius B-15* 1077 700 500 500 15.671 Runout		B- 3	1277	1277	8300	8300	0.361	Ture of Days
B-3 1155 150 7500 7500 10.316 Runout B-3 1462 1462 9500 0.222 Fallure in Test Se B-4 770 770 8000 10.640 Runout B-4 1262 1262 8200 11.37 Runout B-4 1285 1286 8200 12.74 Runout B-5 1286 8200 8200 12.74 Runout B-5 1286 8200 8200 12.857 Runout B-5 1362 1262 8200 9700 0.439 Failure in Test Se B-5 1362 9200 9000 0.439 Failure in Test Se 8 B-14* 1308 8500 8500 0.439 Failure in Test Se 1 B-13* 1308 8500 6500 0.303 Failure at Radius B-15* 170 1077 700 500 5.501 Failure at Small B-15*		, ,	1308	1308	8500	8500	0.258	7657 117
B-3 1462 9500 9500 0.222 Fallure in lest as B-4 770 770 8200 31.137 Runout B-4 1262 1262 8200 31.137 Runout B-4 1262 1262 8200 9000 20.774 Runout B-5 1230 1230 8000 8000 13.657 Runout B-5 1252 1252 8200 8200 10.00 Runout B-5 1353 1353 8200 8200 10.00 Runout B-5 1362 1362 8200 9700 0.439 Failure in Test Se B-6 1362 9200 9700 0.439 Failure at Radius B-7 1363 8500 6500 0.303 Failure at Radius B-14 * 1000 1000 6500 5000 0.508 Failure at Radius B-15 * 1070 1077 700 500 0.508 Failure at R		2-G	1155	1155	7500	7500	10.316	
B-3 120 770 5000 5000 10.640 Runout B-4 1762 1262 8200 31.137 Runout B-4 1262 1262 8200 900 20.774 Failure in Test Seption B-5 1230 8000 8000 32.857 Runout B-5 1262 8200 8200 18.866 Runout B-5 1262 8200 8800 18.866 Runout B-5 1262 8200 8800 10.000 Runout B-5 1363 1308 8500 900 0.439 Failure at Radius B-6 1385 1308 8500 900 0.439 Failure at Radius B-7* 1303 1308 8500 6500 5.61 Failure at Radius B-13* 1308 8500 6500 0.578 Failure at Radius B-13* 1308 1308 8500 6500 5.61 Failure at Radius <td></td> <td>B-3</td> <td>1763</td> <td>1462</td> <td>9500</td> <td>9500</td> <td>0.222</td> <td>ın Tes</td>		B-3	1763	1462	9500	9500	0.222	ın Tes
B-4 1262 1262 8200 31.137 Runout B-4 1262 1262 8200 900 20.774 Failure in Test Section B-5 1286 1286 800 800 18.866 Runout B-5 1262 8200 8200 18.866 Runout B-5 1262 8200 8800 10.000 Runout B-5 1363 1308 8500 900 9700 0.439 Failure at Radius B-6 1385 1308 8500 8500 9700 0.439 Failure at Radius B-7* 1303 1308 8500 8500 2.561 Failure at Radius B-14* 1000 1000 6500 6500 6.501 1.561 Runout B-14* 1000 1000 6500 5000 1.561 Railure at Radius B-15* 107 107 2000 6500 6500 2.561 Failure at Radius		B-3	770	770	2000	2000	10.640	Runout
B-4 1385 1385 9000 9000 20.774 Failure in Test Scaled B-5 1230 8000 8000 32.857 Runout B-5 1262 1262 8200 8200 18.866 Runout B-5 1353 1353 8800 8800 10.000 Runout B-5 1385 1385 1385 1300 9000 0.439 Failure at Radius B-13* 1303 1308 8500 6500 0.303 Failure at Radius B-13* 1000 1000 6500 5000 15.61 Runout B-13* 1000 1000 5000 5000 15.61 Runout B-15* 1000 5000 5000 15.50 Runout B-15* 1000 5000 5500 15.50 Runout B-13* 1308 8500 6500 2.56 Failure at Radius B-14* 1000 1000 6500 5.50		D I	1262	1262	8200	8200	31.137	
B-4 1230 1230 8000 8000 12.855 Runout B-5 1262 1230 8200 8200 18.866 Runout B-5 1353 1353 8800 8800 10.000 Runout B-5 1353 1353 8800 9700 0.439 Failure at Radius B-6 1385 1308 8500 9700 0.303 Failure at Radius B-13* 1303 1308 8500 5000 2.561 Failure at Radius B-13* 1000 1000 1000 5000 0.303 Failure at Radius B-15* 107 1077 700 5000 5.501 Failure at Small B-15* 107 107 700 5500 15.501 Runout B-14* 1060 1000 500 5.501 Failure at Weld B-15* 1077 700 500 0.303 Failure at Weld B-16* 1077 700		12.4	1000	1385	0006	0006	20.774	in Test
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B-5 1253 1353 8800 8000 0.439 Failure in Test Scalus B-5 1353 1362 9700 0.439 Failure at Radius B-6 1385 1385 9000 9000 0.107 Failure at Radius B-13* 1308 8500 6500 2.561 Failure at Radius B-14* 1000 1000 5000 15.671 Runout B-15* 770 770 5000 5000 15.671 Runout B-15* 1077 346 5500 5500 0.578 Failure at Radius B-15* 107 346 5500 5500 15.501 Runout B-13 1308 850C 650C 550 15.501 Runout B-14 1000 1000 650C 550 15.61 Runout B-15 770 700 700 0.338 Failure at Weld B-16 923 600C 650C 550 1	7	8-5	1263	1262	8200	8200	18.866	Runout
B-5 1492 1492 9700 0.439 Failure in Test Scale 1385 1385 9000 0.107 Failure at Radius B-14 1303 1308 8500 0.303 Failure at Radius B-15* 1700 1000 5000 15.671 Runout Feat Scale B-15* 1007 1007 7000 5500 15.671 Runout Feat Scale B-13 1308 8500 5500 15.671 Runout Feat Scale B-14 1000 1000 6500 5500 15.671 Runout Feat Scale B-14 1000 1000 6500 5500 15.671 Runout Feat Scale B-15 1007 1007 7000 6500 6500 15.671 Runout Feat Scale Featlure at Radius B-15 1007 1007 7000 6500 6500 15.671 Runout Featlure at Radius B-15 1007 1007 7000 6500 6500 15.671 Runout Featlure at Weld B-15 1007 1007 7000 6500 6500 15.671 Runout Featlure at Weld B-15 1007 1007 7000 6500 0.338 Failure at Weld B-15 1000 1000 6500 6000 0.338 Failure at Weld B-15 1000 1000 6500 6000 0.338 Failure at Weld B-17 1000 1000 1000 1000 1000 1000 1000 1	Nonwelded	B-5		1252	8800	3088	10.000	
B-5 1434 1325 1385 9900 9000 0.107 Failure at Radius at Rad		B-5	1323	1/02	9700	9700	0.439	בן
B-6 1382 1308 8500 8500 2.561 Failure at Radius B-14* 1000 1000 6500 500 15.671 Runout B-15* 770 770 500 7000 7		B-5	1494	1385	0006	0006	0.107	.at
B-13* 1503 1500 1500 6500 2.561 Failure in Test School B-15* 770 1000 5000 15.671 Runout B-15* 1077 770 5000 7000 0.578 Failure at Radius B-15* 1077 1077 7000 7000 0.578 Failure at Radius B-13* 1308 8500 6500 6500 2.561 Failure at Small B-13 1308 8500 6500 2.561 Failure at Weld B-14 1000 1000 6500 5000 15.671 Runout B-15 770 770 5000 7000 0.578 Failure at Weld B-16 923 6000 6000 0.338 Failure at Weld B-17 923 6000 6000 0.754 Failure at Weld B-18 1155 7500 7500 0.264 Failure at Weld		B-6	1382	- 0000	8500	8500	0.303	at Radiu
B-14* 1000		B-13*	1303	1000	6500	6500	2.561	ב י
B-15* //O 1077 1077 7000 7600 0.578 Failure at Radius B-17* 846 \$500 5500 15.501 Runout B-13 1308 1308 8500 6500 2.561 Failure at Small B-14 1000 1000 6500 6500 2.561 Failure at Radius B-15 770 770 700 0.578 Failure at Weld B-15 1077 7000 0.578 Failure at Weld B-16 923 6000 6000 0.338 Failure at Weld B-17 923 6000 6000 0.754 Failure at Weld B-18 1155 7500 7500 0.764 Failure at Weld		B-14*	1 000		2000	0005	15.671	1
B-15* 197.6 846 5500 5500 15.501 Runout B-13 1308 1308 8500 0.303 Failure at Small B-14 1000 1000 6500 6500 2.561 Failure Outside W B-15 770 770 5000 7000 0.578 Failure at Radius B-15 1077 7000 6000 0.338 Failure at Weld B-16 923 6000 6000 0.338 Failure at Weld B-17 923 6000 6000 0.754 Failure at Weld B-18 1155 7500 7500 0.264 Failure at Weld		B-15*		- 2701	7000	7000	0.578	e_at_Rad
B-13 1308 1308 8500 0.303 Failure at Small B-13 1308 1308 8500 6500 2.561 Failure Outside W B-15 770 770 5000 15.671 Runout B-15 1077 1077 7000 0.578 Failure at Weld B-16 923 923 6000 6000 0.338 Failure at Weld B-17 923 923 6000 6000 0.754 Failure at Weld B-17 923 923 6000 6000 0.264 Failure at Weld B-18 1155 7500 7500 0.264 Failure at Weld B-18 1155 7500 7500 0.264 Failure at Weld B-18 1155 7500 7500 0.264 Failure at Weld B-18 1155 1155 7500 7500 0.264 Failure at Weld B-18 B-18 1155 7500 7500 0.264 Failure at Weld B-18		* *	- -	346	5500	5500	15.501	:
B-14 1000 650c 650c 650c 10000 10000 10000 10000		٦		1308	8500	8500	0.303	at Small
B-15 770 5000 5000 15.671 Runout B-15 1077 700 0.578 Failure at B-16 923 600c 600c 0.338 Failure at B-17 846 550 15.50 Runout B-17 923 600 600 0.754 Failure at B-18 1155 750 750 0.264 Failure at		ر ا	1	1000	6500	6500	2.561	
B-15 1077 1077 7000 0.578 Failure at 1077		, L	<u> </u>	770	5000	5000	15.671	-
B-16 923 923 6000 6000 0.338 Fallure at 13-17 846 846 6000 6000 0.754 Fallure at 13-17 923 923 6000 6000 0.754 Fallure at 13-18 11-55 11-55 7500 7500 0.264 Fallure at 13-18 13-19 10.55 11-55		B-12	1077	1077	7000	7000	0.578	at- - -
B-17 846 846 5500 5500 15.501 Runout B-17 923 923 6000 6000 0.754 Failure at B-18 1155 7500 7500 0.264 Failure at		ی ار	<u>_</u>	923_	9000	0009	0.338	: a'
3-17 923 923 6000 6000 0.754 Failure at 8-18 1155 1155 7500 7500 0.264 Failure at	Melded 	, r	<u> </u>	846	5500	- 5500	15.501	14
8-18 1155 1155 7500 7500 0.2.564 L'Allute ac.			923	923	0009	0009	$-\frac{0.754}{0.000}$	ا ماد
			1155	1.5	7500-	7500	0-264	TOTAL
000			1 Table					

Test Frequency: 30cps HLH FATIGUE TEST DATA FOR ZE41A MAGNESIUM Stress Ratio: R-O Thick Casting Section: TABLE IV.

1 March 1

		Strain Readi Micro-Inc	Strain Gage Reading Micro-Inch/Inch	Applied Bending Stress(psi) $\sigma = E \in \mathcal{E}$	Bending (psi) EE	Number of Cycles to	
Specimen	Number		Alter-		Alter-	Failure	Remarks
Describ		Steady	nating	Steady	119 CT 119	X10°	- 1
	B-7	1385	1385	9000	0006	0.241	Failure in Test Section
	B-8	1308	1308	8500	8500	0.367	Failure in Test Section
	B-9	1230	1230	8000	8000	0.495	Failure in Test Section
	B-10	1155	1155	7500	7500	0.582	Failure at Radius
	B-11	923	923	0009	6000	0.930	Failure in Test Section
Non-	B-12	770	770	5000	5000	0.869	Failure in Test Section
welded	B-21*	923	923	0009	0009	10.027	Runout
_	B-21*	1077	1077	7000	7000	13.715	Runout
	B-21*	1230	1230	0008	8000	0.480	Failure at Radius
	B-22*	770	770	5000	5000	10.041	Runout
	B-22*	1077	1077	7000	7000	0.777	Failure in Test Section
	B-23*	1308	1308	8500	8500	0.327	Failure at Radius
	B-19	923	923	6000	6000	0.300	Failure at Weld
	B-20	1000	1000	6500	6500	0.188	Failure at Weld
	B-21	923	923	6000	6000	10.027	Runout
	B-21	101	1077	7000	2000	13.715	Runout
Welded	B-21	1230	1230	8000	8000	0.480	Failure at Radius
	B-22	770	770	5000	2000	10.041	Runout
	B-22	1077	1077	7000	7000	777	Failure Outside Weld
	B-23	1308	1308	8500	8500	0.327	Failure at Radius
	B-24	1155	1155	7500	7500	0.270	Failure at Weld

*Testing of welded specimen; failure occurred outside weld
 All surfaces of specimen were machined NOTES:

Grain enlargement has usually been associated with zirconium depleted areas by researchers in the magnesium casting industry who have studied alloys containing rare earths such as zirconium and thorium.

Cracking in specimens B-3 and B-6 initiated from surface oriented cavities which displayed no inherent material discrepancies. It is postulated that these impressions/cavities may be associated with the foundry practice of cleaning and coating the molds prior to assembly and pouring operations or from oxidation pits due to incorrect furnace atmosphere.²

Although the predominant mode of failure for thin section specimens was transgranular, fractures in this group were the racterized by a mixed mode and contained both intergranular and transgranular cracks.

Of the specimens representing the thick section of casting, four specimens (B-7, B-8, B-11, and B-12) had fractures which originated at surface oriented pits (Table V). Specimen B-7 displayed large oxide inclusions and grain growth in addition to the surface pit. The remaining specimens, B-9 and B-30 displayed subsurface origins which were located at oxide inclasions. Fracture topography on these six specimens was predominantly intergranular rather than transgranular as observed on the six thin coupons. The microstructure of specimens 3-7 through B-12 disclosed the segragate compound Zn₂Zr₃. This compound usually occurs at high temperatures when the melt is saturated with zirconium. Excess zirconium precipitates and segregates by density difference at the bottom of the casting. This sagregation constituent probably explains the intergranular nature of the fracture surface on the specimens removed from the thick area.

The x-ray quality of these samples revealed numerous oxides and some segregation whereas radiographic inspection of samples from the thin area revealed no discrepancies. Temporary acceptance standards utilizing the thin area radiographs as acceptable and thick area radiographics as unacceptable have been established with the vendor. Preliminary ASTM Ell5 ratings of one and three have been designated for these two conditions relative to oxides and segregation. Definite radiographic standards for magnesium alloys bearing cerium and zirconium should be available from ASTM within 1 year.

Comparison of weld and nonweld fatigue data indicates that welding is detrimental to the fatigue strength of ZE41Z-T5 magnesium. For specimens fabricated from the thin section of the casting, the fatigue strength of the unwelded specimens (Figure 6) is greater than that of the welded specimens (Figure 8). Figures 7 and 9 show a similar trend for the specimens fabricated from the thick section of casting.

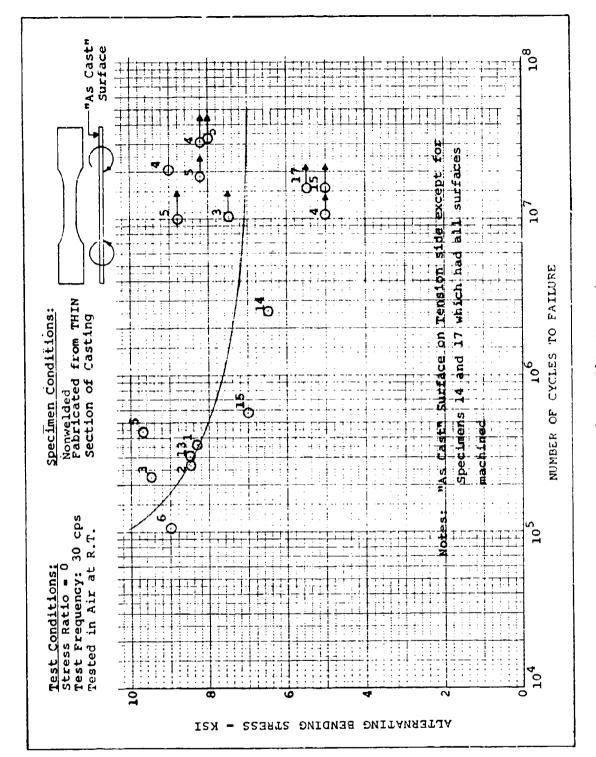


Figure 6. S-N Data for ZE41A Magnesium

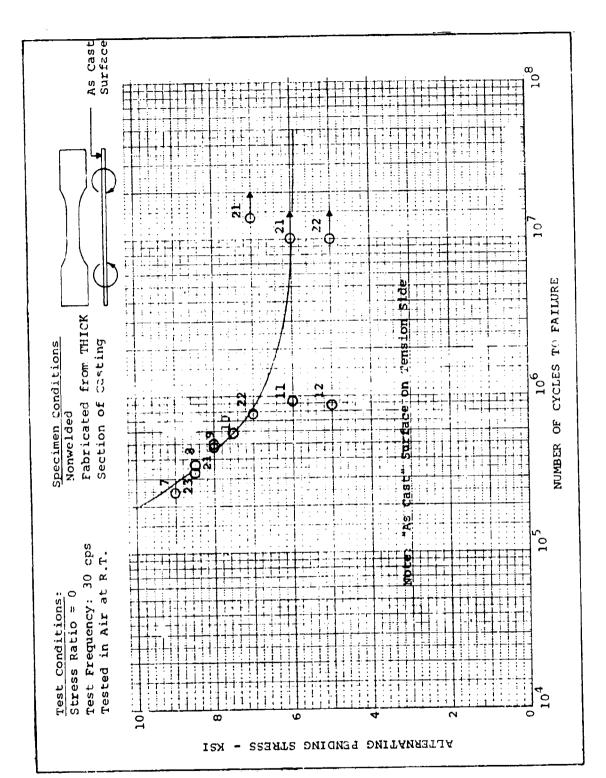


Figure 7. S-N Data for ZE41A Magnesium

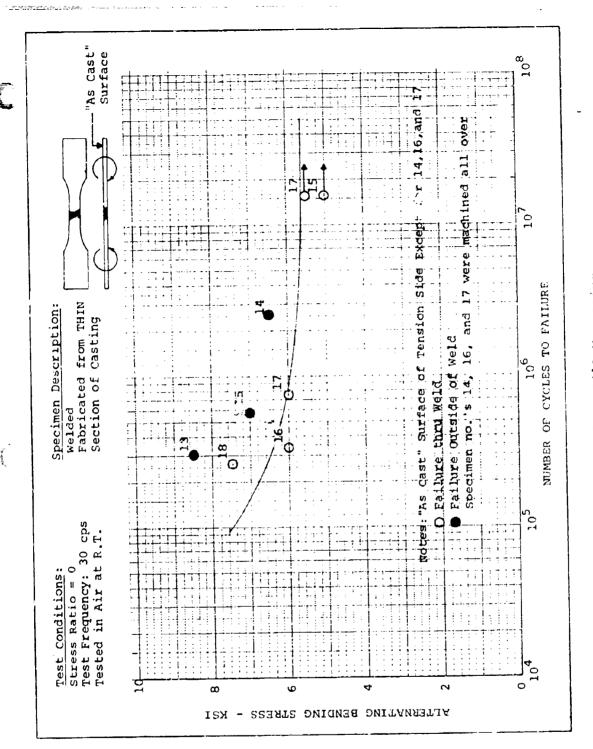


Figure 8. S-N Data for ZE41A Magnesium

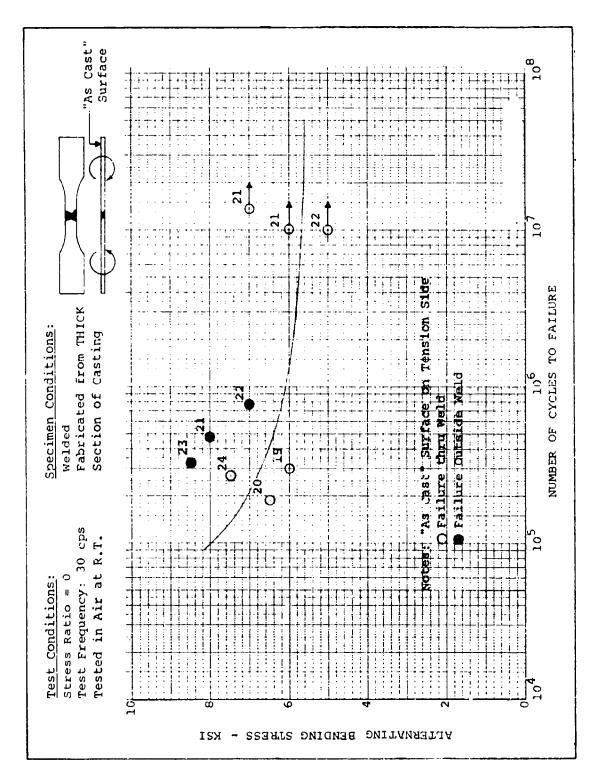


Figure 9. S-N Data for ZE41A Magnesium

TABLE V. RESULTS OF METALLURGICAL EXAMINAT

SPEC- IMEN NO.	FRACTURE LOCATION	ORIGIN LOCATION
B-1	As-cast surface, 1.34" from center	Surface at edge of radius
B-2	As-cast, 1.20" from center	Side surface, .035" from as cast
B-3	As-cast, 0.98" from center	Surface cavity 0.20" from edge radius
B-4	As-cast, 1.26" from center	.010" Below surface, 0.150" from side ed
B-5	As-cast, 0.20" from center	.020" Below surface, 0.490" from side ed
B-6	As-cast, 1.48" from center	Surface cavity, .10" from edge radius
B-7	As-cast, 0.96" from center	Multiple origin from surface cavity & en
B-8	A.s-cast, 0.42" from center	Subsurface, .020" below surface; .030" f:
B-9	As-cast, 1.05" from center	Multiple origin, subsurface; .005" below
B-10	As-cast, 1.72" from center	Subsurface, .010" below surface; .20" fr
B-11	A.s-cast, 0.75" from center	Surface cavity, .220" from side edge
B-12	A.s-cast, 1.30" from center	Surface cavity, .130" from side edge
B-13	As-cast, 0.35" from center	Surface, .010" from edge radius
3-14	Machined, 1.60" from center	Machined surface; .220 from side edge
B-15	A.s-cast, 1.45" from center	Surface, .050" from edge radius
B-16	Machined, 0.10" from center	Subsurface, .020" below surface
B-17		Surface, at edge radius
B-18		Surface, edge of radius
B-19	—	
B-20		Surface, 0.40" from side edge
B-21		Surface cavity, 0.88" from side edge
B-22	As-cast, 1.20" from center	Subsurface, .005" below surface, .38" fr
B-23	As-cast, 1.50" from center	Surface, edge of radius
B-24	As-cast, 0.26" from center	Subsurface weld crack, .050" below surfa

PRGICAL EXAMINATIONS FOR BENDING FATIGUE SPECIMENS

LOCATION	DEFECT SIZE	REMARKS
	.020"Wide x .030"Deep	Entrapped particle at origin; Particle was .050"Long x .010" Thick
s cast	.020"Wide x .020"Deep	Shrinkage or entrapped part.
edge radius	.010 Wide x .030 Long .030 Wide x .010 Deep	Surface cavity
" from side edge	.030 Wide x .010 Deep	Entrapped particle
"from side - >	.030"Wide x .020"Deep	Entrapped Particle
edge radius	.050"Wide x .010"Deep	Surface cavity
ce cavity & entrapped oxide skin	Primary was .050"x.020"	Entrapped particle
rface; .030" from side edge	.040 "Wide x .010 "Deep	Entrapped particle
e; .005" below as-cast surface	Primary was .010"x.020"	Entrapped particle
	Primary was020"x.020"	Entrapped particle
n side edge	.070 Wide x .018 Deep	Surface cavity
rside edge	.100"Wide x .030"Deep	Surface cavity
adius	.050"Wide x .015"Leep	Entrapped particle
om side edge	.020"Wide x .012"Deep	Entrapped particle
radi us	No visual defect	
ırface	.10" Long	Weld defect; weld fracture
	No visual defect	Weld defect; weld fracture
	No visual defect	Weld fracture
s-cast surface	0.0502"Long x 0.030 "Wide	Weld fracture
e d ge:	No visual defect	Weld fracture
n side edge	.010"Wide x .010"Deep	Surface cavity
orface, .38" from side edge	.020"Wide x .020"Deep	Entrapped particle
	.020"Wide x .030"Deep	Entrapped particle
00" below surface	.038" Long	Weld fracture

approximately 50 percent of the cases where failure occurred in the weld zone, the origin of failure was found to be at an area of porosity or other defects. Side-by-side photographs illustrating the microscopic fracture characteristics plus the origin location and microstructural details of the origin sites are displayed in Figures 14 through 17. Of the six specimens representing the thin section of casting, three specimens (B-13, B-14, and B-15) fractured outside the weld bead zone. Specimens B-21 and B-23 had oxide particles at the origin while B-22 contained no apparent material discrepancy.

A valid comparison of fatigue strengths of ZE41A-T5 and AZ91-T6 could not be made. No bending fatigue tests of AZ91-T6 were conducted in this test program. Previously obtained data on AZ91-T6 was questionable with respect to several important test parameters and was unsuitable for making a sound technical comparison of the fatigue strengths of the two alloys. However, the fatigue testing of the ZE41A-T5 material indicates the current fatigue design allowable stress of ±1000 psi currently used for AZ91C-T6 transmission housings is applicable and is conservative for ZE41A-T5.

FATIGUE CRACK PROPAGATION

The primary objectives of the fatigue crack propagation testing were:

- 1. Compare the fatigue crack growth characteristics of ZE41A-T5 with AZ91C-T6.
- Determine if weld repair is detrimental to fatign crack growth.
- 3. Determine if a difference exists in the fatigue crack growth characteristics of material removed from thick and thin sections of castings.

Test Specimen Configuration and Preparation

Thirty-two fatigue crack propagation specimens, 16 of ZE41A-T5 and 16 of AZ91C-T6, were fabricated from large size castings as shown in Figure 1. For each alloy, specimens were fabricated from material removed from thick and thin sections of the castings. Both unwelded and welded specimens were tested. The proportions of these compact tension type specimens were generally in accordance with the recommendations of ASTM Test Method E399-70T. The welded specimens were cut from the casting to orient the weld at the center of the crack starter notch (see Figure 1) such that the crack would tend to propagate within the band of weld metal.

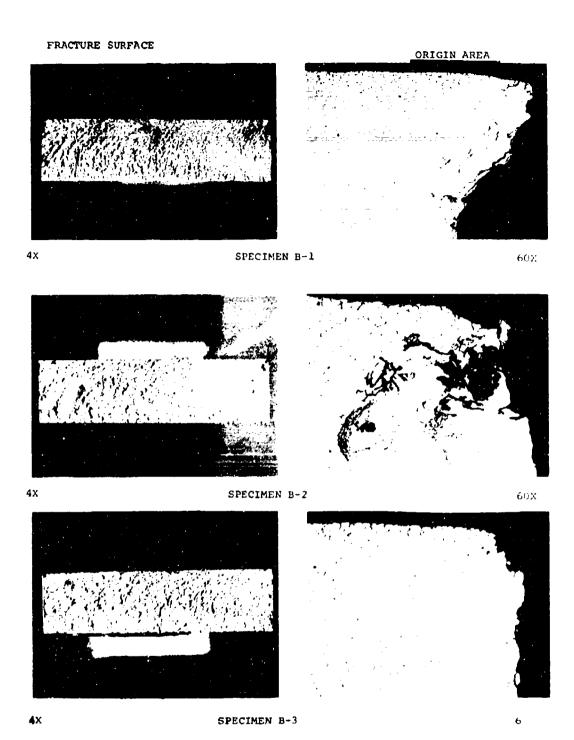


Figure 10. Origin Location and Microstructure Fatigue Test Specimens B-1, B-2, B-3

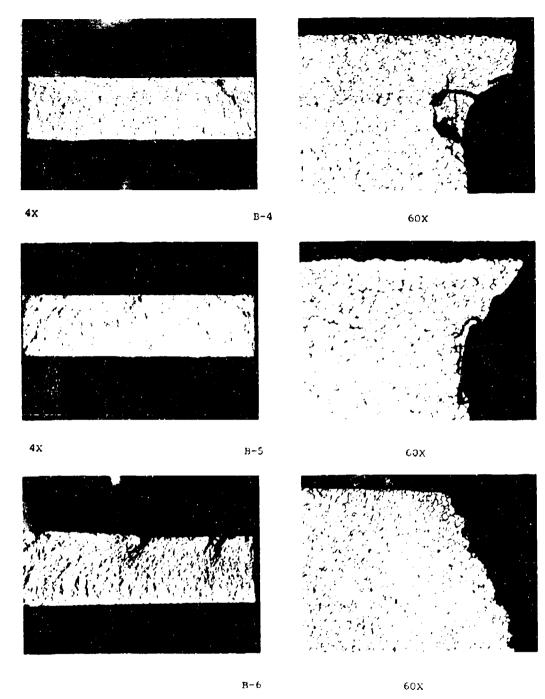


Figure 11. Origin Location and Microstructure of Fatigue Test Specimens B-4, B-5, B-6

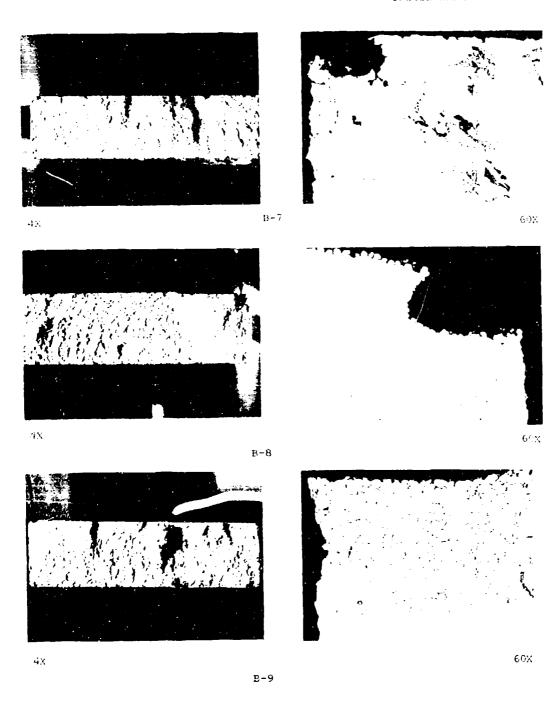


Figure 12. Origin Location and Microstructure of Specimens B-7, B-8, B-9

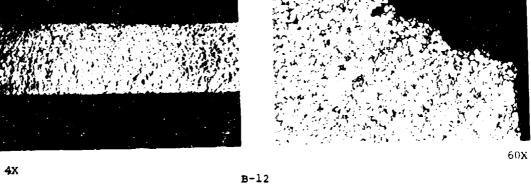


Figure 13. Origin Location and Microstructure of Specimens B-10, B-11, B-12

Figure 14. Origin Location and Microstructure of Welded Specimens B-13, B-14, B-15

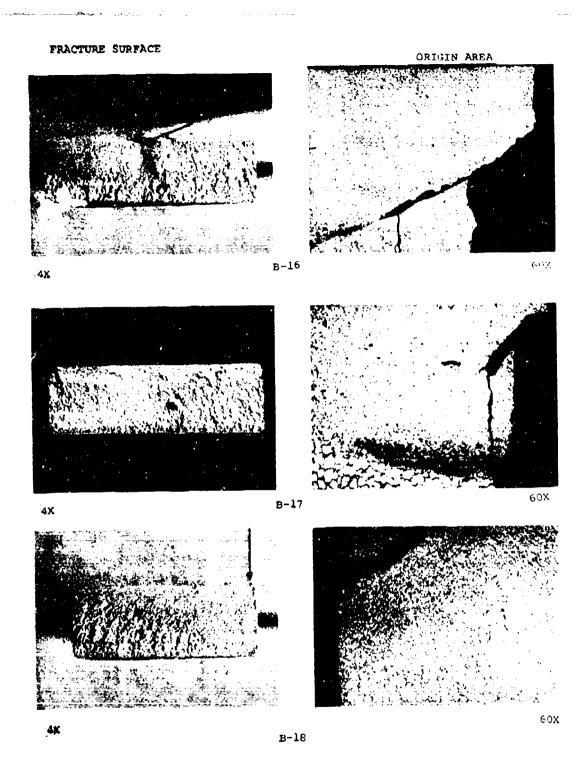


Figure 15. Origin Location and Microstructure of Welded Specimens B-16, B-17, B-18

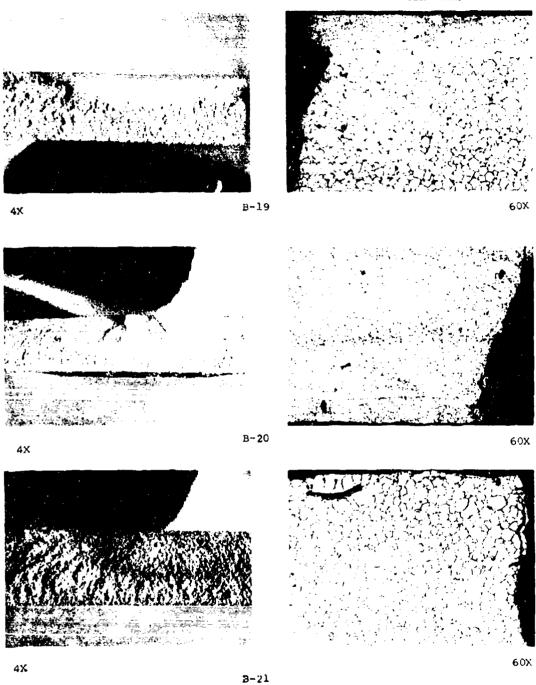


Figure 16. Origin Location and Microstructure of Welded Specimens B-19, B-20, B-21

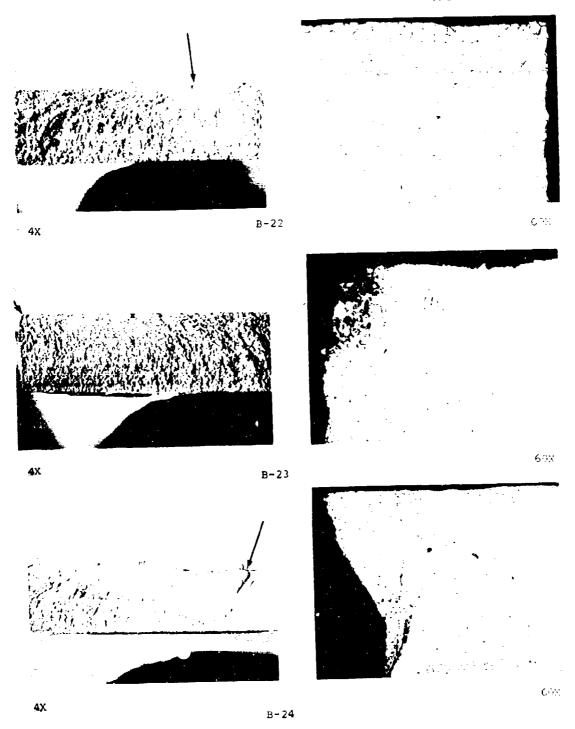


Figure 17. Origin Location and Microstructure of Welded Specimens B-22, B-23, B-24

Details of the material processing, including welding, are discussed in the <u>Material</u> section.

In order to measure crack growth, a grid consisting of approximately 40 lines with a nominal spacing of 0.040 inch was photographically applied to one side of the specimens. A typical specimen with the grid installed is shown in Figure 18.

Test Procedure and Test Setup

Table VI contains a matrix of the fatigue crack propagation specimens. Because of machining errors, 27 specimens were tested rather than 32 as originally planned.

The fatigue crack propagation tests were conducted in air at room temperature. A stress ratio of 0.714, representing the anticipated relationship of steady and alternating stresses in service, was used in these tests. Loads were applied at 10 Hz in the test setup as shown in Figure 19. The test machine contained a load cell in series with the specimen. Load control was provided to permit no greater than ±1.5 percent variation of the cyclic range of load for the duration of each test. In cases where precracking loads higher than the test crack propagation loads were required, care was taken to step down to the test loads in small increments and to let the crack grow to a length such that the prior load would not influence the crack growth data.

Crack growth was monitored visually by observing the intersection of the crack front with the grid lines previously described. Dye penetrant (type MIL-I-25135, Spotcheck SKL-HF Penetrant) and optical magnification of various power (15X and 45X) were used as aids in following the crack. Periodic checks were made to insure that cracking was progressing uniformly on both sides of the specimen. In no case was more than 0.04 inch difference in the crack lengths observed from one side of the specimen to the other.

Basic crack growth data, consisting of crack length and number of cycles, was analyzed using the techniques of fracture mechanics. A computer program was used to calculate the fracture mechanics parameters of stress intensity range, ΔK , and crack growth rate, Δ a/ Δ N. Tables VII and VIII present the data which were used to prepare Figures 21 through 28. The stress intensity range was calculated using the following expression which is found in Reference 3.

$$\Delta K = \frac{\Delta P}{BW^{1/2}} \left[29.6 \left(\frac{Q}{W} \right)^{\frac{1}{2}} - 185.5 \left(\frac{Q}{W} \right)^{\frac{3}{2}} + 655.7 \left(\frac{Q}{W} \right)^{\frac{5}{2}} - 1017.0 \left(\frac{Q}{W} \right)^{\frac{7}{2}} + 638.9 \left(\frac{Q}{W} \right)^{\frac{9}{2}} \right]$$

where Δ P is the load range, $P_{max} - P_{min}$

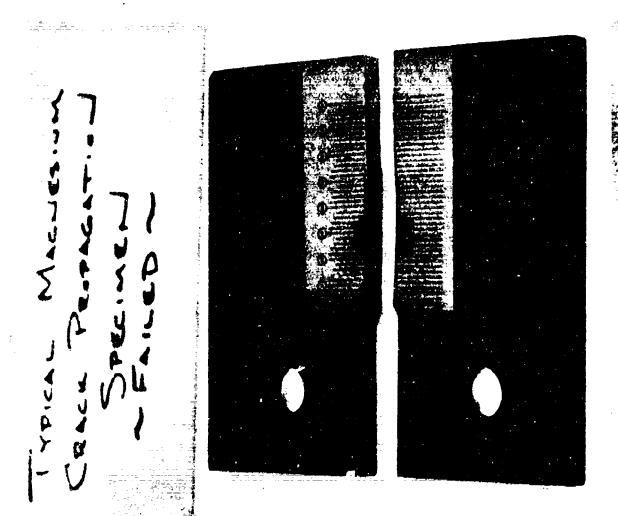


Figure 18. Crack Propagation Specimen with Grid Installed

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S CONDITION REMARKS	Nonwelded	4			_			Nonwelded	Welded	-					- >-	Welded	Nonwelded	4 —		Nonwelded	Welded	•	-	Welded	Nonwelded	•	~•	Nonwelded	Welded	•	Welded
CASTING SECTION	Thin	Thin	Inin	utul	Thick	Thick	Thick	Thick	Thin	Thin	Thin	Thin	Thick	Thick	Thick	Thick	Thin	Thin	Thin	Thin	Thin	Thin	Thin	Thin	Thick	Thick	Thick	Thick	Thick	Thick	Thick
FATIGUE CRACK PROPAGATION SPECINENS (CASTING SUPPLIER CASTING SUPPLIER CASTING SECTION)	Vendor A								Vendor A	Vendor B	Vendor A	Vendor B	Vendor B	Vendor B	Vendor A	Vendor B	Vendor A	4 -							-						Vendor A
VI. SUNDARY OF FA	ZE41A-T5														-	ZE41A-T5	AZ91C-T6	-4													AZ91C-T6
TABLE PECIMEN DESIGNATION	11	C 5		C 4							7	_	7	~	~	Н	_	-	-	7	7	7	7	7	~	7	~	7	7	M	C 31 C 32
S	1					~~~										36	5										_				

(1) Not tested, machining error (2) Reduced size specimen, shown

Reduced size specimen, shown in Figure 29

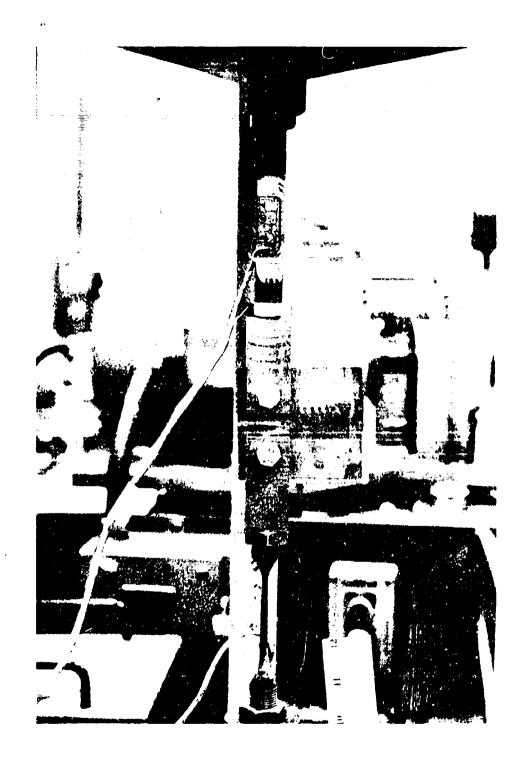


Figure 19. Crack Propagation Setup

Figure 29 indicates the relation of the above dimensional parameters to specimen geometry. All fatigue crack propagation specimens except C-10, C-12, and C-14 were fabricated to the geometry shown in Figure 1. Specimens C-10, C-12, and C-14 were of the geometry of Figure 29.

Test Results

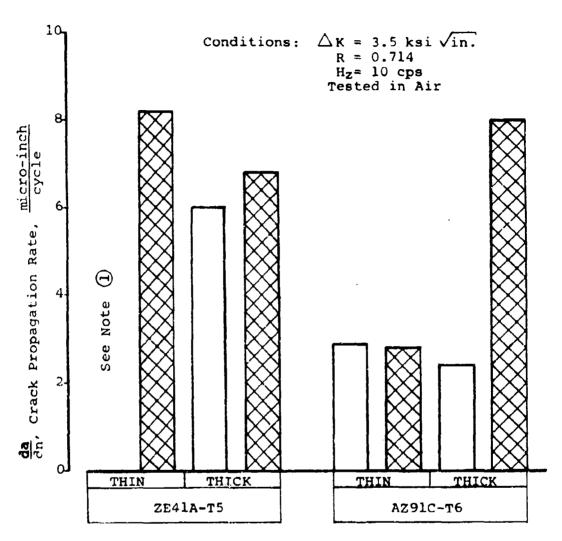
The results of the fatigue crack propagation testing are summarized in Figure 20 which shows the relative crack growth rates for various conditions of the two alloys. Detailed data are contained in Tables VII and VIII and displayed as plots of stress intensity range versus crack growth rate in Figures 21 through 28.

For material in the unwelded condition, AZ91C-T6 (Figures 25 and 26) exhibited slower fatigue crack growth than ZE41A-T5 (Figures 20 and 22). This difference is attributed in part to differences in the microstructure of the two alloys. It should also be noted that yield and elongation values obtained for the AZ91C-T6 material in this program are not typical of average values obtained by testing samples cut from actual transmission housings. Yield values determined in this program were lower while elongations were higher than those typically obtained from actual transmission housings.

For the ZE41A-T5 material, no difference was observed between the crack growth rates of welded (Figures 23 and 24) and unwelded material (Figures 21 and 22). In the case of the AZ91C-T6 casting, the welded specimens (Figure 28) utilizing material from a thick section showed a faster crack growth rate than the unwelded material (Figure 26). No significant difference was seen in the crack growth rates of welded (Figure 27) and unwelded (Figure 25) AZ91C-T6 material removed from thin sections of the castings.

The ZE41A-T5 material showed no significant differences in the crack growth rates of specimens removed from thick (Figures 22 and 24) and thin (Figures 21 and 23) sections of the castings. Unwelded AZ91C-T6 specimens exhibited the same trend.

In all cases, the fatigue crack propagated normal to the applied load. For the welded specimens, the crack remained within the band of weld metal, and metallurgical examination indicated that the band of weld metal extended completely through the thickness of the specimen.

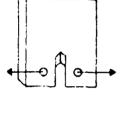


NOTE ①: Limited data at this condition does not warrant a comparison

LEGEND:	
	Nonwelded
	Welded

Figure 20. Crack Growth Comparisons for Magnesium Alloys at a Constant Stress Intensity Value

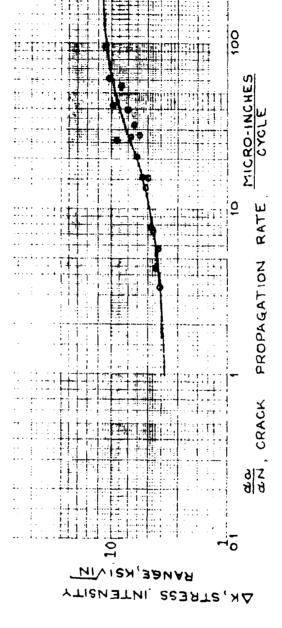
LOADING FREQUENCY 10 COS
ENVIRONMENT: AIR
TEMPERATURE: R.T
SPECIMEN LOCATION: THIN SECTION



CASTING

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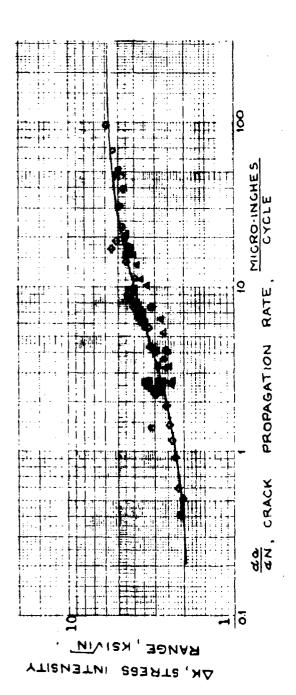
STRESS RATIO 0.714 LEGEND: O SPECIMEN G-1



Danigue Uruck Growth water for DEATH Wagnesium Casting Crack Propagation Testing of Nonwelded Specimens tager 21.

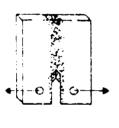
0 LEGEND STRESS THICK SECTION CASTILLE LOADING FREQUENCY: 10 CPS A'A T P SPECIMEN LOCATION: ENVIRONMENT TEMPERATURE:

C-8 0 C-6 2-5 SPECIMEN SPECIMEN SPECIMEN SPECIMEN 4110 RATIO $\triangleleft \square \Diamond$



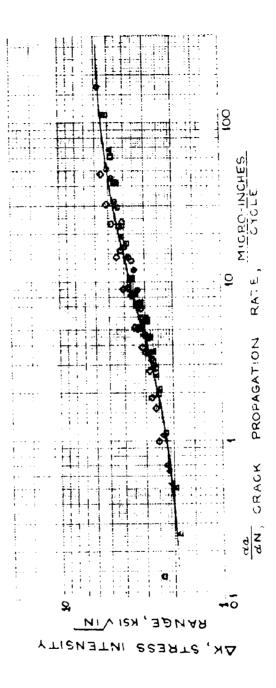
Fatigue Crack Growth Rate for ZE41A Magnesium Casting Crack Propagation Testing of Nonwelded Specimens Figure 22.

LOADING FREQUENCY: 10 CPS
ENVIRONMENT: AIR
TEMPERATURE: R'T
SPECIMEN LOCATION: THIN SECTION
OF CASTINA



CTRESS RATIO 0.714
LEGEND: O SPECIMEN C-9

\$\times \text{SPECIMEN G-10}\$
\$\text{O SPECIMEN G-10}\$
\$\text{O SPECIMEN G-11}\$
\$\text{SPECIMEN G-12}\$



Felloud Neat Crusts For ZB411 Meguesium Cepting Crack Propagation Testing of Welded Specimens را د

LOADING FREQUENCY 10 CPS

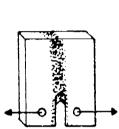
ENVIRONMENT:

TEMPERATURE:

SPECIMEN LOCATION THICK SECTION

OF CASTING

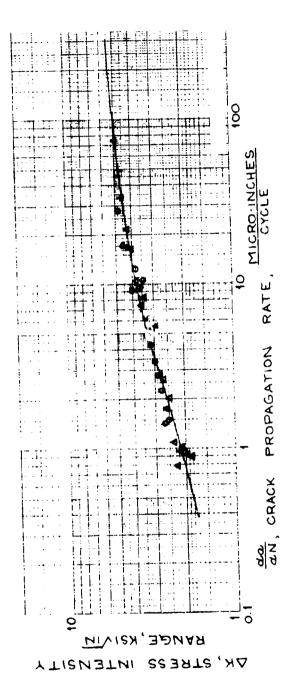
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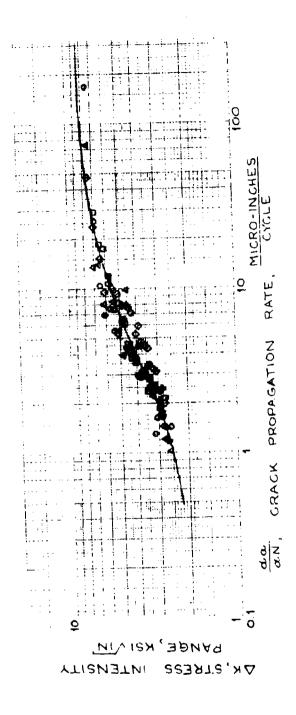
STRESS RATIO O.714

LEGEND O SPECIMEN C-14

\$\times \text{SPECIMEN C-15}\$

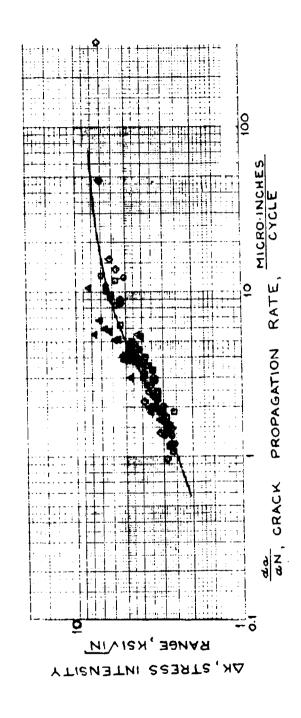


Fatigue Crack Growth Rate for ZE41A Magnesium Casting Crack Propagation Testing of Welded Specimens Figure 24.



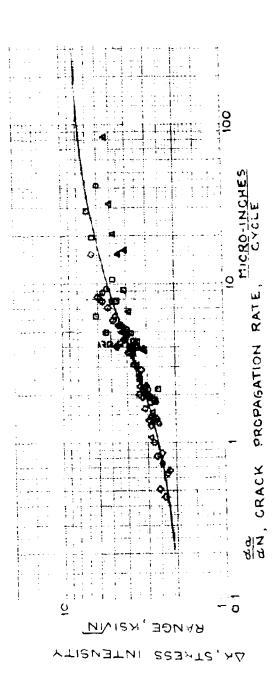
Estigue Crack Growth Rath for AZ917 Magnesium Casting crack Propagation Testing of Nonwelded Specimens 52 odiália

C-26 C-27



Fatigue Crack Growth Rate for AZ91C Magnesium Casting Crack Propagation Testing of Nonwelded Specimens Figure 26.





Patigos Crack Good? Rath [ch 82%]; Magnesium Casting Crack Propagation Testing of Welded Specimens Fichte 77.

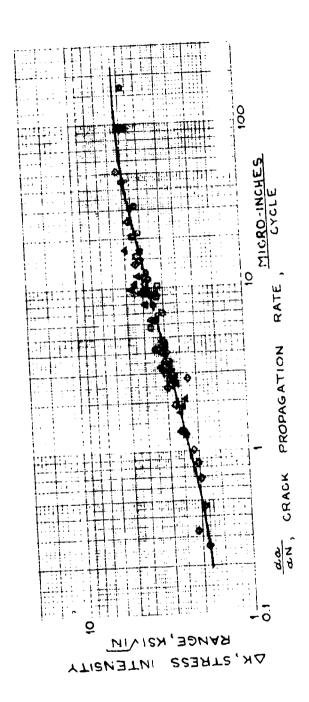
ENVIRONMENT:

R.T

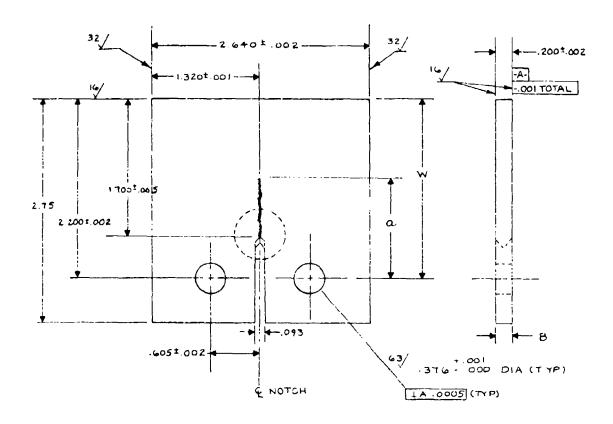
SPECIMEN LOCATION: THICK SECTION

OF CASTING





Fatigue Crack Growth Rate for AZ91C Magnesium Casting Crack Propagation Testing of Welded Specimens Figure 28.



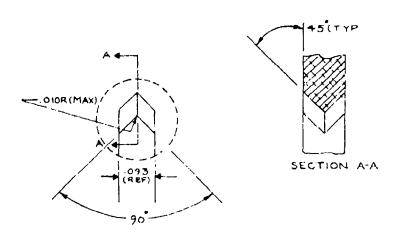


Figure 29. Modified Crack Propagation Specimen for ZE41A-T5 Welded Magnesium

NOMENCLATURE - TABLE VII and TABLE VIII

The following nomenclature is applicable to the crack propagation data contained in Tables VII and VIII.

P_{MAX} = Maximum applied load (steady + alternating), lb

P_{MIN} = Minimum applied load (steady - alternating), lb

A = Initial crack length measured from centerline of holes to line at which first test reading was taken, in.

N_O = Initial number of cycles corresponding to crack length, A_O

B = Specimen thickness (Reference Figure 29), in.

W = Distance measured from centerline of holes to back of specimen (Reference Figure 29), in.

A = Crack length measured from centerline of holes to crack tip, in.

N = Total number of applied cycles

DELA = Change in crack length, in.

DELN = Change in applied cycles

 K_{MAX} = Maximum stress intensity factor, psi \sqrt{in} .

DELK = Stress intensity range, psi Jin.

DADN = Crack propagation rate, in./cycle

TABLE VIII. POPULIE OBACK PROFACKO TON CARL POU VIKING DASHO ON LAND AND MACAMETUM CASHO ON LAND

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TABLE VII - Continued

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۲.	ë	13614).	3.4)))))F-	011 456).		4231.64	0.877195-05
	õ	138063.	0. 40000F-	01 1920.	_	4456.92	0.238136-04

TABLE VII - Continued

WELDED: THICK SECTION OF CASTING: R=0.714: 10 CPS

C-7:2641A:NCN-WE	NCN-WELDED: UHI	LULUSTHICK SECTION OF CASTINGS CONTROL OF CONTROL				
	P. M. A. X. 4. 20.	PMIN 300.	An 0.99700	0 ° 0	8 0• 19960	2.50003
	;	e C	N - 30	X A A	DELK	DAON
			-	10117.78	2 690, 80	0.33303E-05
307 FO !		0.4000E-0			3012 78	0. 23981F-05
307701	238)).	0-30(00%°C		C / • + + C []	0.0100	70 27 27 20 0
111706		0.430306-0	1 28680.	11 100. B4	3.43.10	00-1444 C-00
10111		0 400005-01		11489.64	3282.75	3. 25 991E-JS
11570E		5 10000 ·	, , , , ,	12715 80	3433.09	0.25247E-05
11570E	37540	0. 40000F-U	-,	00.000		20-146415
307561		0.40001F-0	1 5580.	12585.00	119666	CO-150001.0
10101		C-3010020		13234.11	3772.63). 65354E-J5
12117	•	0-30000	1 4020	13881.35	3965.10	0.995028-05
131 70F		0. 40000r - 0		60 90 7	4170 99	0.158735-04
13570E	105780.	0.40000E-0		14.00.00		20-32-031-0
307061		0.403305-3	1 2520.	15440.42	71.4144	0. 136/35-04
36 / 601				15363, 36	4675.25	0. 3921 bf -04
14370E	109350.	0. 430000 - 0	4		91 7707	0. 51282F-34
Ë	.001011 10	C. 40000F-0	1 740 :	1 181 . 04	01.006.	

TARLE WIT - Continued

C-R:25411:NON-WEIGEDITHICK SECTION OF CASTING:28-0-715:13 Con

	X & ∑ X	MIM	۸Ö	C 7	8	38
	α	275. 0	•) .	156	2.5 0000
<:	2	4 190	21.0	X Z Z Y		DACA
7452 JF		3.430305-01		7110,61		0.51580E-06
78520F		0,400005-01	981.60	1179.69	16.8605	0.437508-06
32 × 2 0 F		C. 40) 33 F-31		7594.63		3. 591028-06
84520E		0, 400365-01		7955.83		0.777338-36
306206		0.400005-01		2151,15		0.115347-35
3(2500		0.41111E-11		8467.19). 14 19 45 - 1) 5
105256		6.433005-31		40.00 m		0.139315-15
.10252 c		0.400005-01		9165.27		0. 253506-15
13652F		0.41)))F-);		9547.33	2728.37	90-344146-08
.11352E		0. 400JOF - UI		9350.27	2845.51	0.23392E-09
.11452F		0.4000CF-01		1)394.27	2973.93	0. 22))26-35
118525		0. 40300F-31		10.07.01	3105.72	0.239816-05
.1225F		0.400305-01		~	3251.31	0.550226-05
12652		0.40))05-11		•	34)9.42	J. 61 16 2E - 35
.13052F		0. 40000F-01		12537.29	3582.08	0.653596-05
365411		0,400305-31		ċ	3771.67	0.69444E-05
138525		3.403335-11			3980.94	0.74905£-05
14252F		0. 40001E - 01			4212.99	0.952405-05
353541.		0.40000E-01			4471.29	J. 141845-34
.:505;		C. 40000E-01		~	4759.71	0.22383F-04
1565251		0.400005-01		~•	5082.66	0.190446-04
158525		3,43)305-11	2343.	19388493	5444.83	3, 17,3946-54
7	*69068	ı	4.20°	30430° 05	5.851.41	0.672338-04

TABLE VII - Continued

	PMAX	ZIEG	₽0	02	æ	¥
	0	350.	0.74810	°.	0. 20190	2 • 5 00 3 0
⋖	Z	DELA	DELN	KMAX		DADN
78810F		0. 4000E-	13500.	9250.28	2642.94	0.29630E-05
82910E		0.40 JOSE-		9555.72		3. 32 84 1 E - 35
86810F		0. 40000F-		3913.15		0.44743E-05
90 A 1 OF		0.4C000F-		10282.41	2937.83	0.58997E-05
94 P 1 3E). 4) 30 9F-		13481.98	3351,99). 666675-05
988 I OF		0. 40000F-		11108.97	3173.99	0.74074E-05
102818		0.4C030E-		11554.28	3304.08	C. 87719E-35
13681F		3.4)))9E-		12343.68	3442.77	0.11905E-04
110815		-300004-0		12567,92	3590.84	0.1333E-04
11491F		0.40000E-		13122.89	3749.40	0.14184E-J4
118815		0° 40000E-		12-10-11	3919.92	0.15657F-04
122A1F		0.40000F-		14364.46	4104.13	0.23809E-04
12681E		0.40)30E-		15364.94	4304.27	0, 28985E-04
130816		0.40000F-		15830.27	4522.93	0.31746E-04
u	01 74169.	0.40000F-01	01 900.	16670.94	4763.13	0.45454E-04
13881F		0.400306-		17539.15	5028.33	51 28 2E - 0
142816	31 75300.	0. 43000E-		18529.72	5322.49	
		20000	070	10775,00	5650.02	7. 16667F-13

C-13:2E41A:WELDED:THIN SECTION OF CASTING:R=0.714:13CPS

	PAAK 715.	PM IN 225.	4() 0.68480	ぐ ● アウ	B 0.20120	2.20000
4	Z	AHU	S THE	7. 2. 4.	DELK	NOVO
	15828).	0. 40000F-0	154	5552 88	1872.25	0.252726-0
	238080	0. 49000F-3		5821.01	1948.86	0.50125E-0
	13320.	0.40))08-)		7113.93	2 132 55	0-531638-0
	375960.	0. 400000 - 0		7431.07	2123.16	0.63857E-J
	4332.10.	0.400ccr-a		2777. 74	2220.78	0.498816-0
	473; 13.	3.40))Jf-]	1 369)).	214).12	2325.75	J. 1)8435-0
0.96480F 00	489120.	0.40300F-U		A535.57	2438.74	0.210318-0
	504780.	0-400001-0		g 162.53	2560.74	0.255436-3
	515823.	3.43030F-3		3425.88	2693.11	0.36232E-0
	527160.	0. 40000F - 0		0031.75	7837.64	0.35273E-0
	535923.	C. 40000F-0		1)497.93	2996.55	J. 45 66 2E - J
	541320.	0. 430308-0		111104.00	3172.57	0-340146-0
	546060.	0-40030F-0		11791.05	3368.87	0.84368F-0
	5499)).	3.40))JE-J		12552.36	3589.16	0.134176-0
	552720.	0.40000F-0	1 2820.	13432.10	3837.74	0.14184E-0
	554823.	0.400006-0		16417.91	4119.40	0-190436-0
	555793	3,49931F-J	1 96).	15514.71	4439.63	0-389915-0
	556380.	0.400006-01	1 600.	15915.73	4804.50	0.556575-0

TABLE VII - Continued

C-11:2E41 A:WFLDED: THIN SECTION OF CASTING:R=0.714:1009 S

	•	PMAX 420.	*00E	0.74420	A0 •420	0°0	8 0• 20220	2.50000
۵		z	DEIA		DELN	KMAX	DELK	NOVO
20E	00	34800.	0.40000E-	0.1	34800.	7892.11	2254.89	0-114946-0
. 82420F	Ç	321313.	0.40333F-31	-01	286533.	8159.62	2331,32	J. 13962E-0
. 96420F	00	363660.	0.40000E-	01	42360.	8452.01	2414.86	0.94429E-0
.90420F	00	384000	0.4C030E-	10.	20340.	8768.30	2505.23	0.196666-0
. 9442 JF	2	3991)).	3.4)) 2) F-01	.01	14100.	9133.04	2402.30	0.28369F-0
.98420E	00	410880.	0.400005-	.01	12780.	9471.16	27:36.35	3. 31 2995 - 3
.13242F	10	410760.	0, 43330F-	-31	8880.	9358.40	2816.69	0.45046E-0
.10642E	10	426720.	0.40000F-	01	6940.	13271.26	2934.65	0.57471 E-0
. 11042E	01	433443.	0.40330F-	.31	6723.	13711.96	3063,56	0.59524E-J
.11442E	:0	441300.	0. 40000E-	10	7860.	11193.78	3195.36	0.53891E-0
11842E	5	446943.	0.400006-01	0 1	5640.	11690, 79	3340.23	0. 70922E-0
. 12242E	1	451623.	9.430336-	.31	4683.	12239.41	3496.69	0.854705-0
.12642F	01	455460.	0. 40 300E -	0.1	3840.	12332.96	3666.56	0-17417E-0
.13042E	0	457830.	0.40C00E-	0.1	2340.	13482.20	3852.36	0.17394E-0
.13442E	11	459630.	- 300006 -	.01	1800.	14134.94	4055.70	0. 22 22 2E-0
.13842E	01	461400.	0. 4.3000E-	01	1800.	14981.63	4280.46	0-322550
.14242E	01	452360.	0.400)0F-	.01	96).	15853.73	4529.63	0. 41 66 7E - J
. 14642F	01	463020.	0. 40000F-	01	660.	15824.51	4807.00	0.63636E-3
.15042E	0.1	463280.	0.40000E-	10	360.	17909.27	5116.65	0.111116-0

C-12: 2E41 A:WELDED: THIN SECTION OF CASTING: P= 0.714:10 CPS

		PMAX 420.	PH IN	46 (3.5866)	40 166)	25 . (8 3. 231 33	2.2.3333
❖		7	4 12 (1)		OFLN	X A B X	DELK	NOTE
ANE	00	40543	0.403)36	-)]	4,156.).	8751,23	2533.21	3. 485146-36
76660F	00	655AU.	0, 400006-01	-01	25020.	9120.65	2605.90	6.159875-05
ROKADE	000	87010.	0.43000F	10-	21420.	9513.04	2718.01	0.135746-05
84540F	? =	1314)).	0.43))E	-) [144)).	9937.75	2839.36	J. 27775F-05
88660F	00	112560.	0. 43300F	-01	11150.	10395.25	2970.07	0.35842E-05
92 5 6 0 F	80	1228F0	0.400308	-0:	10320.	10387.25	3110.64	C. 397608-15
9646.JF	<u> </u>	133833	3.43033E	-)1	7920.	11414.75	3261.93	0.50505E-05
100665	;	138723	0.40000f	- 01	7920.	11988.68	3425.34	0.53536E-05
10466	· -	142580.	0 40000 E	-01	486).	12639.38	3632.68	0.823045-35
10866E		147960	0. 40000E	-01	4380.	13287.39	3796.40	0.91324E-05
11266F		152460	C. 40000F	-01	4500.	14033, 10	4009,46	0.838886-05
	·	155043	0.403335-31	-)1	258).	14859,26	4245,53	3,15534€~3¢
12066F	; ;	157860	0.40000F	-01	2820.	15780.80	4503.80	0.141846-04
12466F	50	159600	0.40030F	-01	1740.	16915.37	4804.39	0.22988E-04
1286F	; ;	16.1923.	9,40,3395-31	-);	1320.	17383.03	5138.01	0.303036-04
13266E	. 6	151760.	0.400006-01	-01	840.	19306.37	5516.11	0.476195-04

TABLE VII - Continued

C-14: ZE41 A: WELDED: THICK SECTION OF CASTING: R=0.714: 13CPS

	_	PMAX 420.	PMIN 3006	A7 0.68510	0.0	B 0. 23033	2.23333
4		z	DFLA	DELN	KMAX	DELK	UADN
0.72510F		25700.	0.40000E-01	01 26700.	8777.72	2537.92	J. 14 981E-35
0. 7651 JF		55020	0, 40000F-01	01 28320.	9137.00	2610.57	0.141246-05
3.80510F		72960.	0. 4.0000E-	17940.	0450,50	2722.71	0.22297E-05
0.84510F		97333	0.400)0F-	11 1434).	56. 455t	2844.11	J. 27894E-J5
0.88510F		99240	0. 4C000F-01	-1	13412.23	15.7162	0.335016-05
0.92510F		109720.	0.40000F-	01 9480.		3115.56	0.42194F-U5
3.96510F		116983.	0.403705-			3266,95	3.490205-05
0.10051F		120660.	C. 40000F-	3780.		3430.41	0.10582E-04
0.104516		124863.	0.40000F-			3607.79	0.952386-05
0.10851E		1284)).	0.40000E-	3540.	13305.08	3801.45	0. 11 299E-04
0.11251E		132780.	0. 40000E-	01 4340.	14050.52	4014.43	0.91324F-05
0.11651F		135190.	0.40000F-	01 24.33.	14876.16	4253.33	3. 16667E-34
0.120515		137520.	C. 4 3000E -	01 2340.	15736.95	4513.41	0.17094E-04
J.12451E	10		0.4000E-	01 1440.	16830.47	4808.71	0.277786-04

C-15:2841A:WFLDED:THICK SECTION OF CASTING:R=3.714:13705

		⋖	Ž.,	40	Č.	.	्र तर भ
		350.	•	J• 75 J+)	-	69767 •6	orrec.
•		7	RFLA	K 1 3 0	XAIX	0 P.F. K	PADW
1 0		, d	1 -31 (17 7)	11 65123	54.26.25	100 * 5 X II	J. 8455 3E-JC
10,706,01		• · · · · · · · · · · · · · · · · · · ·			כת רנגי	15.029.	0.938758-06
0.93040E		30 J F O.	0.400001-0			• • • • • • • • • • • • • • • • • • • •	30-3146 80 O
30405A C		131820.	0.400005-01		16 746	60.617.7	
10. 91. 74.75		1710)).	J. 49))))E-		7334,07	2395.12	00-160701 00
0.000.00		212040	0,403005-01		7520.50	2117.30	0.951326-06
10,000		C69846	0.400008-0		73,54.66	2264.43	0. 747)95 - 76
10 to to to		3)3663.	1). 4))))f = (4250.91	2357.40	0.138766-05
370 401 1		320500	0. 40000F = 0		8497.75	2456.50	U. 23 78 JE - 75
0.01.0		203.636 263380	0.40000F-		9768.36	2562.30	J. 17544E-JS
110066		358620	0.400005-0		9184.74	2675.65	0.25247F-05
110076		372770-	0.4000F-		9701.30	2797.52	0.28359[-05
3 40 40 40		230267), 40110F-		13252,41	2929.26	0,563226-35
340001 00		387240	C. 40000F-(10752,55	3072.45	0.54200£-05
10 17 10 10 10 10 10 10 10 10 10 10 10 10 10		20206	0.400005		11301.21	3555.65	0.53605 6-0 5
0.1351.0	7 7	3087)).	0.400306-31	4863.	11932.87	3400.82	0.82304F-05
0.139046		404100	0. 4000CF -		12567.47	3540.71	0.740748-05
370671 0		407940	0.400001-		13334,83	3801.37	3, 134176 - 34
167065		410400.	0, 40000F - 01		14125.92	4035198	0.15.26.0F 04
7701=1		412260	- 30000+ 0		15043, 15	4598.64	0.215556-54
			-301105-0	_	16.13.1.52	4591.58	3, 33,33,35, 3,4
1,000					16	£ 25.5.5.7	40 3540-250
0-10904	Ē	マンウンオーネ		•			

TABLE VIII. FATIGUE CRACK PROPAGATION DATA FOR ZE41A and AZ91C MAGNESIUM TESTED IN AIR

C-17: AZ91C: NON-WELDED: THIN SECTION OF CASTING: R=0.714:10 CPS

	X 4	× 4 9 0 •	PMIN 350.	AO 0.70800	0 7 0	8 0. 20130	2.50000
~		z	0F1 A	OFLN	¥ A	DELK	DADN
. 74 P G	00	28020.	0. 40000E-0	_	8991.29	2568.94	LLI.
7880	00	62133	3.40)))5-3	(7	77.1	2653.63	11 73 7E
8280	00	90120		1 28020.	9593.41	2740.97	1+276F
8580	00	114600.	0.40000E-0	1 244	9938.73	2839.64	15340€
0900	((138060.	7.43013F-3	1 23463.	13312.37	2946.31	17050F
9480	00	162720.	0.40000E-0	مس	10717.73	3060.78	15 22 1 F
986	00	193500.	0.40000E-01	1 30750.	11140.95	3183.13	0.12995E-35
1028	10	2.05.02.0.	0-401101-0		11537.55	3313.59	34722E
1068	0	216840.	0. 40000E - 3		12084.33	3452.67	33841E
.1108	10	229380.	0.40000E-0	_	12604.00	3601.14	31 89 8E
1148	010	243360.	0. 4000UE-0	_	13153.69	3760.20	28612E
.1188	01	256850.	0.40000F-0		13759.03	3931.15	29630E
.1228	21	267243.	3,43330E-7	1 13383.	14475.62	4115.89	38536F
. 1268	010	277080.	0, 40000F-0	1 9840.	15108.00	4316.57	30 59 C5
1308	010	286380.	0.40300E-0	1 9300.	15875.46	4535.85	430116
1348	16	292563	0.430308-3	1 6180.	15718.42	4116.69	64725E
.1388	0.10	330120.	0. 40000E - 0	1 7560.	17649, 13	5042.61	52910E
1428	01	307443.	0.40001F-0	1 7323.	18681.55	5337.59	3949K
1448	01	312720.	0. 40000E-U	1 52	19431.05	20.9995	75757E
.1508	01	318480.	0.43000E-3	1 57	21114. 84	6632.81	9 4 4 4 E
1548	01	322320.	0.4)0)0E-)	1 384	22552.33	6443.52	134176-3
43.51.4	0.1	2339	0.4000F-0	1 1	24154.82	6904.23	256418-0
	6	2412	0.400301-7	1 2:00	28975, 02	7421.43	0.156576-03

SERVICE SERVICE AND AND SERVICE AND SERVICE SERVICES SERV

	:	:	,	57	(£	3
	* 00 %	200	71.73.3	· ·	0. 20143	2.503.3
<;		45	: * : :	अ ब ५ ४	O.E.E.K	7047
754000	30000	40000	1054).	٠ بر	2582.99	3, 1311 65-35
0.7000407.0		0,400,105-0;	134 43.	3331,95	2565.27	80-37421 C
300468	•	400000-3	33130.	53.0	758.2	: 05 :
שליי אלם "		43333-1	2773).	F (358.4	66847
3004.c.	•	4.100004	13300.	C.	365.t	23643
10000	•••	4. 3005-3	154:3.	a	382.4	15:12
שלו ייפס		433330-3	·(riel	٠.	4.502	22002
367:0.	,	6-100012	*C7 [7]	.•	332,5	55606
*0376°		400004	141.1.	اب ب	7	5949)
111605	,	400006-0	12540.	.1	529.7	31 595
000011		40000-0	17000.	a,	293.9	31.746
EC9011		-311102	1392).	ur E	964.3	36.03.)
30986		4.00006-0	10500	~. ~.	152.0	38095
-12760-	•	400000-0	8750.	45	356.0	45652
-6366		V - FOCCO 4	47.87	۲.	579.0	68056
308681		40000-0	10260.	Ś	324.5	38 9R 6
30904.	•	4000004	7250.	٦,	395. A	55 396
1000	•	400000	.4300	7.5	395.9	33333
はつかんしょ		- 300003	3350	٣.	732.3	\$0677
	••	-3010E-1	*080*		197.2	33.323
104516	•	- 20(007	1197	iri	5,77.3	0.8673
はなくと、	•	- よりかさなす	•036	15 56+56	+ . 10 c é	5031
					C	*************************

TABLE VIII - Continued

C-19: AZ91C:NON-WELDED: THIN SECTION OF CASTING: R=0.714:10 CPS

1611-151-151-15))	;				
	PMA	AX	N EN	AD.	40	C ?	80 %	78 u
	52	5.	375.	0. 70	930	. .c	0. 20160	000000
<		z	OEL A		DELM	XAYX	DELK	NOVC
74080F	CO	19630.	40000E	-01	96	532.	2752.06	0.21505E-0
78980F	000	41520.	300005		000	9939.51	2~39.86	0-174528-0
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. 1149AE	31	175980.	0.43030E		11283.	14101-47	4030-71	0.35461E-J
- 118 38F	01	186720.	0. 40000E	-01	10740.	14749.80	4214.23	0.37244E-0
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TABLE VIII - Continued

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	• • • • • •)	* * * * * * * * * * * * * * * * * * * *	• • • • • • • • • • • • • • • • • • • •	्र केट्ट इ.स. १८ वर्षा	3. 68 9.88 8F - 14

TABLE VIII - Continued

C-23:AZ91C:WELDED:THIN SECTION OF CASTING:R=0.714:10 GPS

	4		(A)	CZ	80	ĸ
	400	350.	0.70580).	0.19980	2.50000
•	2	V 140	2 1 2	X A M X	DELK	DACIN
A 77 6 0 35		3, 40000F-01	18123.		2585.91	0.22075E-05
30,007		0.40000E-01	21900.	37.	2667,90	0.18255E-05
300076		0-40000E-01	2922)		2758.73	0.13689F-35
86690E		0. 4.3300F - 0.1	22320.	10002.53	2851.87	0.179216-05
A OF	111900	0.400005-01	20340.	10377-82	5865.09	C-19656E-05
94687c		3.43333F-31	23593.	13783.64	3080.18	J. 19436E-05
98480F		0. 40000F - 01	13240.	11211.25	3203.21	0.219336-35
10268F		0.40006-01	15960.	11670.42	3334.41	0.25063E-35
10668F		2, 4)))JE-)1	9240.	12150.93	3474.27	0.43290E-05
11068F		0. 400000 - 01	9240.	12687.45	3623.56	0.43290E-05
11468F		0.4C300F-01	8243.	13241.98	3783.42	J. 48 1 J 9E - J 5
11868F		0, 40000F-01	8400°	13443.56	3955.30	0.476198-05
12268F		0. 40000F - 01	4440.	14493.42	4140.98	0. 90 090 E-05
126685		3.4)))06-)1	954).	15199.36	4345.68	0. 41 92 9F - 05
1306AF		0. 40030F-01	5100.	15370.38	4 562 .96	0.784316-05
13468F		0.40000F-01	3720.	16817.23	4804.92	0.10753E-04
12XA8F		3,40))36-)!	9723.	17752.39	5072.03	0.41152E-05
14069E		0-401006-01	9043.	18788, 91	5368.26	0.43751F-05
16668		0.4CCOUE-U1	9660.	19943.26	5698.07	0. 41 4 38E - 35
1506AF		0. 40000F-01	6420	21232.45	4.9	0. 62 30 5E-0 5
1546RE		U. 40000F-31	2040.	ď.	478.8	9608E
15868F	1 261933.	3.400008-01	134).	24294.61	6541.32	28 48 5 6 - 3

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	XVWd		AG.	C _F	æ	3
	423.	200.00	0.71160		0.20130	2,50000
<:	.7	< 1 €	N 150	2537		7040
7516 35	50873.	3. 4000UF-01	100 x 50°	7-11,35		6.553076-05
791605	122500	6. 40000F- 31	62700.	TALL DE		3.63736F-PA
0. 04150.0	12 ABC 11	11-11-1007 0	0724).	53.6 10.4	2340455	6. 45 50 7F - 3 6
071405	257220.	0, 400 JOE - 01	46860.	16 16 16 16 16 16 16 16 16 16 16 16 16 1	2445.57	0.853516-04
909116	376376	0. 40 300F-01	78900.	V	2537.75	0.504971-06
96,636	185.17).	0.4)1)15-11	4893).	15.8666	2636.63	0, PJ ROOF -06
509166	4745304	0. 463308 - 31	344.93.	66.4020	2742.26	6.131328-05
103166	455340.	0.400004-01	71860.	9105016	2854.90	0.125556-05
177156	425227), 4111116-11	02002	.)	HC 765	0.130716-35
1111165	510120.	0.43000F-31	24940.	10861.28	3103.22	0.151336-05
11 51 6F	504040	0.400005-01	1392).	11342.34	3241.58	3. 29734E-35
119166	544020.	U. 43000f - 01	1999).	11859.21	3388.35	0,200201-05
123168	556920.	0.40C00F-01	12900.	12414,32	3546.09	0.31 009 6-05
127165	567643.	0.40333E-31	1)74).	13325.05	3721.73	0.37244E-35
13116E	578460.	0.40000F-01	10,400.	13590.18	391105	0.37037E-35
13-16	*025785	0-100000+0	9066.	14420.14	4120.04	0.44150E-65
1391651	504793	0.403398-31	726).	15225.42	4350.41	J. 55 39 6E - 0 5
191651	604560.	0. 40000F - 01	4780.	15121.16	4606.04	0.479308-05
391491	611345.	0.40000F-0	47 AG.	19.1117	4890.74	3.58497E-35
151166	61 7046.	0. 400007-31	5700	18259,03	520F . F.C.	6.701756-05
155166	622200	0.400015-01	5150,	10477. A1	4565.69	0.775215-05
391551	627313.	3.40))))[-)[480).	37876. JG	5564.60	0.833338-35
	£2064C.	0.40000 01	2560.	2255h. 36	72.5123	0-1251010

TABLE VIII - Continued

C-25:A291C:NON-WELDED: THICK SECTION: OF CASTING:R=0.714:10 CPS

C-25: A 241 C: NUN-HG	C. NON!						
	P. A.	× .	PMIN 350.	A'J 0.71440	0 · 0	в 0. 20223	2,50000
				2 . 90	X	DELK	NO A O
ø		Z	DELA	00.00	·	2560.54	0-11926E-05
		13540.	0.400005-0		٠	10.0000	3 1236 15-15
. 754	0 (-40000	_	•	2652.24	U. 15 50 JE - 3.5
•	00	0 2 4 4 7 0				2743.53	0.11884E-05
0. R3440F	00	97140.	10000F		99 0500	7843.04	0.151176-05
	00	123630.	+0000E-		•	2950.46	133
9710	33	15363).	3.43333E-3		•	2046.47	0.136081-05
7750		174000	6. 400005-0		•	200760	7777
10 4 4 6 6 0 0 0 0 0 0 0 0 0 0 0 0 0 0 0		193660	C-40000F-3		•	2100-10	70
*	37	216060	0-40000F-0	_	•	16.6166	
10344	7.	• C C O O I Z			•	1459,80	23.5
10744	01	233160.	4 JOOCE -	.	•	3639.19	24 5
_	10	5 69650•	+30033F-	.	•	1769.27	25.99
0.11544F	10	264240.	0. 43000F-0	1	•	1941.53	297
0.11944E	01	277680.	40000F-	_	•	6127.79	374
0.12344F	1	288363.	41)))6-		•	42.056	629
0-127445	01	296880.	43000L-	01 8520.	•	4551 81	42.1
, ,	0)	306363.	40000E-	-4	٠.	4795.29	9.0
, -	31	314880.	4)	_	•	5064.36	123
	01	318120.	4.3000E-		٠.	5362.98	854
14344	01	322800.	40000E-		•	5695.61	833
14744	10	32.7630.	40000F-		•	6067-40	92.5
15144	01	331920°	40000t	7	* ~	6483.76	0.13929E-34
.1554	11	355R	3.4)JJ1E-)l	C w	01 *6 6077	4950.94	0.12579E-04
15944	10	338760.	4 JOOOE -	r)	_) 1	

CLEATING OF ASTRONOS ASTRON <u>6-26: A291(:: h. . . 71 (E111)</u>

	PMAX 490.	PN174	۸۵ 0×667.0	\$. 0	H 0.19337	2.50000
<	-	\$ 4 U.C.		* * *) (1) (2)	20 KG
70520F		0-500005.0		24.54.40	2701.17	0.11204E-05
940176		1 4 3 3 3 3 5 - 3	283.23.		2794,83	Sir Byolatic
F7630F		0 400001 0		C3 : 2 : 10 .	2895.74	30-386251.0
9396E		0-300005-0		:3453°:5	3006.69	0.139475.05
3(c 5 5		3, 433336-3			2124.52	0.155436-35
102650		0.40)306-3		61.544.1	1250.31	97 321126-38
350501		0-400306-0		11:000	3 284 . 42). 23575F=34
177035		0.400000-0		1,7364.04	3527.45	0.18414F-05
111936		0.0000000		12443.	3:30.23	0.19 4411 05
11593E		0.403335-3		13454.19	3844.35	J. 31 44 7E-J5
110011		0.400JOF-J		14711-41	4020.40	0.537676-05
326221		0.400008-0		16730, 35	4211124	0.54645E-05
127936).43113F-)		15455.13	4413.89	0.43290E-05
13193F		0.40000F-U		16251.39	4646.11	0.294958-05
16295		0 12900F 0		12177 61	62 6275	3 53 546F · 5
147076		0. 40000E - J		23374.54	5821.30	0.51680€-0°
366151		0. 40001F-3		21712.75	6203.64	0,574738-0
155070		0,40000		2321: 58	6 e 3 l . 0 l	0. 59524E -05
		0. 43.300F 0		76.56.55	7112,55	0,65657E-05
16534		0 - 400004 - 0		26.547.25	7652.10	0.550065-05
F. 167.03F	33263	0 433335-31		12 /26000	8257.92	3, 19 59 2E - 0.4

TABLE VIII - Continued

C-27:AZ91C:NON-WELCED:THICK SECTION OF CASTING:R=0.7:4:10 CPS

	_	PMAX	ZI	00	C Z	60	70 00 70 10
		•067		0, 70933	•	05 6 5 1 • 0	0.0.004.2
ď		2	DELA	DELV	XAMX	DELK	NUAC
7493	00	21423	0. 40000E-0	~	9393.43	2597.27	0.186745-05
	00	58320.	0-433006-0		9380.07	2680.02	0.13840F-05
8293	3	92523	3.40-J2 JE-)		9733,58	2771.59	3.11696E-05
8693	00	115830.	C. 40000E-01	1 232.80.	13050.29	2871.51	0.171826-05
9093	00	134640.	0-40000f-0		10428.30	2979.51	0.17637E-35
9493	Ç	150420.	J. 4))))f-1	1 20940.	10833.89	3095.40	0.191025-05
£600	00	175680.	0-400006-0	1 15250.	11257.30	3219.23	0.24633E-05
1029	01	18 3543.	0.40000F-0		11729.45	3351.27	3. 28 86 JE-25
1069	10	201540.	0.43000F-0		122221	3492.04	0.3333F-05
	0	212880.	0.43000E-01	11340.	12.84.21	3642.35	0.35273E-05
1149	7	222910.	0.433338-3		13311.65	3603,33	3, 3992.36-35
1189	10	234000.	0.400/JOF-J		13917,52	3976.43	0.350366-05
1229	5	242943.	C. 40000E-0	1 8940.	14572,30	4163.52	0.44743E-05
126.9	01	253623.	0.43501F-3		15293.82	4366.80	0.37454E-05
1309	10	261720.	0. 40000F - 0		16051.23	4588.92	0.49383E-05
1349	01	272223.	0.40030E-01	1 10500.	16915.39	4832.97	0.38395E-35
1389	=======================================	282480.	0.400005-01	1 10260.	17458.48	5102.42	0.389865-05
1429	0.1	248780.	0.40000F-71	6.3	18304. 79	5401.37	0.63492E-05
1469	01	292233.	3.43))))E-)1	1 342).	20)69.98	5734.28	0.11696E-34
1509	10	296460.	0.400001-0	1 4260.	•	0.90	0.93897F-05
J.15493F	0.1	303423.	0.40000F-01	1 3960.	22428.82	4522.52	0.10101E-04

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			• • • • • • • • • • • • • • • • • • •	3.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7.7		2226246	3.30 3.00 3.00 3.00 3.00 3.00 3.00 3.00
∢.		27	لا الله الله		x 4:. x	2) 21 31	2040
	CC	L -	0.433335-31	*5.5522 11	18 9383	3673.82	3 14.55 ot - 35
· ·	00	,,	C, 40000F-3		3774115	1776.08	153616-3
, CF 12 A	0	L1	C. 2.3000E- 0		6.44.0	7 885.57	95.7851-6
00000000000000000000000000000000000000	, ,	u .	3,4333)()		. .	102.41	0-309651
8701		.,	2, 40000015		337.	126.45	5-4405 85
1088	C	- (0.400007-0		14041	23.04.5	74.36.753
α		-	0.4)))))[-1		1336	3401.86	0 -31 DE CE
X 20 10 10 10 10 10 10 10 10 10 10 10 10 10	ני	٠ <u>,</u> ۲	0. 40 300E - 3		<i>* ((((((((((</i>	3584.25	777111-0
1208	0	0.7	0.40000F-0		3)16.	2718.53	21 535F-3
0.174835	0	21960C.	0. 40000f - J			5846.49	0.317661-05
1288		. ~	0.43990F-3		4316	4060.36	37075-0
3 2 5			3.400016-3		. 580.	4302.78	399216-3
α 4 ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° ° °		7	0. 40000F - J		S878.	4536.72	45 97 75 - 0
1408			D 100001		1324	1.755,07	0-305144
1448		u U	0.40 JJJE-J		, 1837.	5083.45	C-319105
1400		-	0. 40000F - 0		6915,	5404.45	88 88 8 E-0
9631		1~	0.400008-0		0171	5763,36	138846-0
a + 3 - 1 -		7	0, 40000F-0		0881	6165.8C	158776-0
46.7		~	0 40000E 0		3160.	e617.30	47519E-0
6741 *		7	0.403336 3		10 A 15 5	1124.53	33333F-J

TABLE VIII - Continued

0.78431E-05 0.12121F-04 0.16667E-33 0.65658E-03 0. 399238-35 0.35273E-05 0.31299F-05 3, 92 59 2F - 0 5 0.112996-04 0.11494E-04 0, 21 535E-34 0.317455-04 0.303038-04 0.27548E-05 0.35088E-05 2.50000 3779.32 4138.77 4341.84 4563.89 4837.96 3328.79 8469.03 3618.83 3952.02 DELK 2850.67 2958.38 3073.89 3197.27 2750.92 0.20167 16827.88 2665.90 5973.63 13227.62 4485.71 15196.43 13354.23 0754.62 11193.44 11553,77 2141.60 3832,06 0977,36 X Y Y X 9528.22 SECTION OF CASTING: R=0.714:13 CPS 9 1520. 5100 24). 3300. 3480. 1963. 1260. 11340. 4223. 3540. 3323. 12780. 14529. 11400. DELN 0.79430 0. 40000F - 01 0,400306-01 0. 40000F-01 0.40)))[-)1 0.40000F-91 0.400006-01 0.400008-01 U.40000F-31 0. 40001F-01 0. 40000F-01 3.43335-31 3. 40000E-31 0.400135-11 C. 40000E-01 C. 400006-3 DELA PM IN 350. C-29: AZ91C: WFLUFD: THICK 84540. 73023. 79830. 81650. 84483. 84540. 8202B 60060. 64283. 69480. 76320. 1452) 5020. 35¢40. 47240. 490 PMAX 10010 01 01 10 10 10 CO 00 0.90430F 0.11543F 0.11943E 0.10747F 3.131436 0.13543E 0.11143E 0.12247F 0.91430F J. 1 J? 4 3F 3.127435 0.139435 3.874305 30:438.6

C-30: AZ91C:WELDED: THICK SECTION OF CASTING: R=0.714:10 CPS

	PMAX 456.	PMIN 325. 0.	A7 0.71060	5 °	8 0. 20153	₩ 5.50030
•	7	DF 1 A	2-13-0	XMXX	DELK	0 A D V
	- 0	0.400004-01	-	08 3565	2387.40	0.208336-05
75050F		10-100000-0		8627. P4	2453.67	0.187275-05
70.36.3E	40100		71360	KO. 810 K	2548.01	0.17139F-05
83060F		10-100004-6			2640.00	0.25 24.)5-35
87040F		0.400001-11			00.0000	2 240116-05
91)6 JE) 0366).	0.433336-31			7 * * 6 % 1 %	20 31 10 2 0
960AC30	105260.	0.400)01 - 01		62.1566	/ 84h • U8	0.341336.03
90000		C. 43C30F-01	9)6).	1,125,18	26.0.052	0, 44 [5 Jt = 05
3000		0. 40000E-01		13785.34	3041.54	0.38 98 6 F - 35
17706		0. 40000F - 01		11214.73	3211.07	0.328411-05
		0.430315-11		11722.93	3349.41	0.641)4E-35
306.111		10-100004 0	6420.	12241.52	3497.58	0.62305F-05
115051		0 100001 0		12799.20	3656.92	0.43492E-05
100711		10-300007		11432.09	3829.17	0.77519E-05
127 JAF		16-10/10/14-0		14757.23	4016.35	0.02592E-05
12705F		0. 43700E = 01		200 1100	75 (66 7	0. 11 696F - 34
13:06F	1 168723.	0.4C000F-01		07 • 6 7 • 5 1	F	0 105025
13536F	1 1725.30.	0. 40000F-01	3780.	15-50-03	21.6544	FO-120C01 •0
1200FF	176640.	0.40000f - 31		16427,03	4694.01	0. 4561 41-05
14305F		0.400105-11	2345.	12363-21	4464.44	*(-1446) 1 *(-

- Continued TABLE VIII

3332.53 3638.76 3393.68 4096.93 2883.44 2982.59 3207.37 3760.73 3923.03 0.19917 13817.39 XAAX 3730.59 2633.65 10045.05 0439.05 1663.84 3162.56 4339.27 C-31:AZ91C:WELDED:THICK SECTION OF CASTING:R=0.714:13 CPS \mathcal{C} 57.33. 3900° 4560. 13560. 14.14). 4200. 4260. 2400. 4760. 0,75360 0. 40000F-01 C. 400)3E-31 0. 83333F-31 0.43030F-01 3. 40 JJ JE - 01 U. 40000F-01 0.40000F-01 0. 43000F - U1 0.400306-0 DEL A PMIN 375. 14753. 29320. 37920. 54983. 673RD. 34023. 42480. 5652). 60720. PMAX 525. 00 00 0 0 0 <u>..</u> 0.11536E 0.11936E 0.913505 J. 79767E 3.83360F 0. 87360F 3.95360F 0.10336E 0.10736F 0.11136F

0. 27 100F-05

2.50000

3. 731756-35

0.29498E-05

0.87719E-05 3.55 980E-05 0.95238E-05 0, 93 83 7E-05 0.15667E-04

C. 10256F-04

0.95238E-04 0.95233E-04

0.15534E-04 3. 23202E-34

4486.35

4706.35

15472,23

4284.04

4994, 15 5732.21

2580.

198).

0. 40000F-01

71940.

72360.

•09669

0

0.12336F 0.127365 3.13136F

0.49300F-31

0, 40000F-01

0. 40000F - 01

TABLE VIII - Centinue?

C-32: AZGIC: MCLDEP: THICK SECTION OF CASTING: P.D. 714:10 CPS

	⋖	213	िंद	C _Z	S	ł
	250.	250. 3.	3.71363	·.	3. 23123	2.53333
•	2	0.61.4	7740	X	DELK	2000
75.06.0F		^	15144).	£634.14	1839.47	0.264136-36
76050		0	Ġ.	5543. R3	1899.24	0.45493F-36
RACKOF		(7)	53540.	6471.27	1963.22	0.683058-06
87.36.35		1,43)1,15	47643.	7116.34	2034.10	3. 3396 1E - 3 b
91040F	`		12~200.	7247.65	2110.70	J. 32 206 F-06
95040F	-	0.400001-01	34943.	7475.00	2192,83	0,102445-05
96 76 96	•	0,411)11-01	14400	41.52.6	2290.07	0.276428-05
1340561	,	0,419306-91	*00 #C6	8213.35	2374.30	0.124516-05
10706	•	0.400001-01	30120.	8-,503,25	2474.13	3.132435-35
11106F	_	0.430015-01	3 JL 20.	3337.43	5580*98	C. 13291F-05
11506	_	16-466994-91	21050.	Q2 1 1 5 2 Q	2694.85	0.180036-05
11976	• •5	0.433335-33	1244).	9451.68	2817.62	J. 31.153F-35
12306F	•	0.430004-31	15200.	13,26.20	2950.34	0.246915-05
127065	•	0.400305-01	8450.	10430.09	1604.57	0.47291E-05
131 765	. •	0,400)05-)1	94.30	11332.73	3252.20	0.42194E-05
135065	·		9540.	11094.89	2425.40	0.419296-05
133065	•	0	4330.	12554.44	3616.73	0, 91 3246-35
1420AF		Ö	3840.	13401.33	3828.95	C. 134176-04
14706F	•	0.400JUE-01	2769.	19.82671	4065.32	0.144938-04
151 985	•	\circ	2893.	15152.82	4320.38	0.13889E-04
155066	•	0.400005-01	1560.	_	4625.08	0.25641E-04
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Enclosures

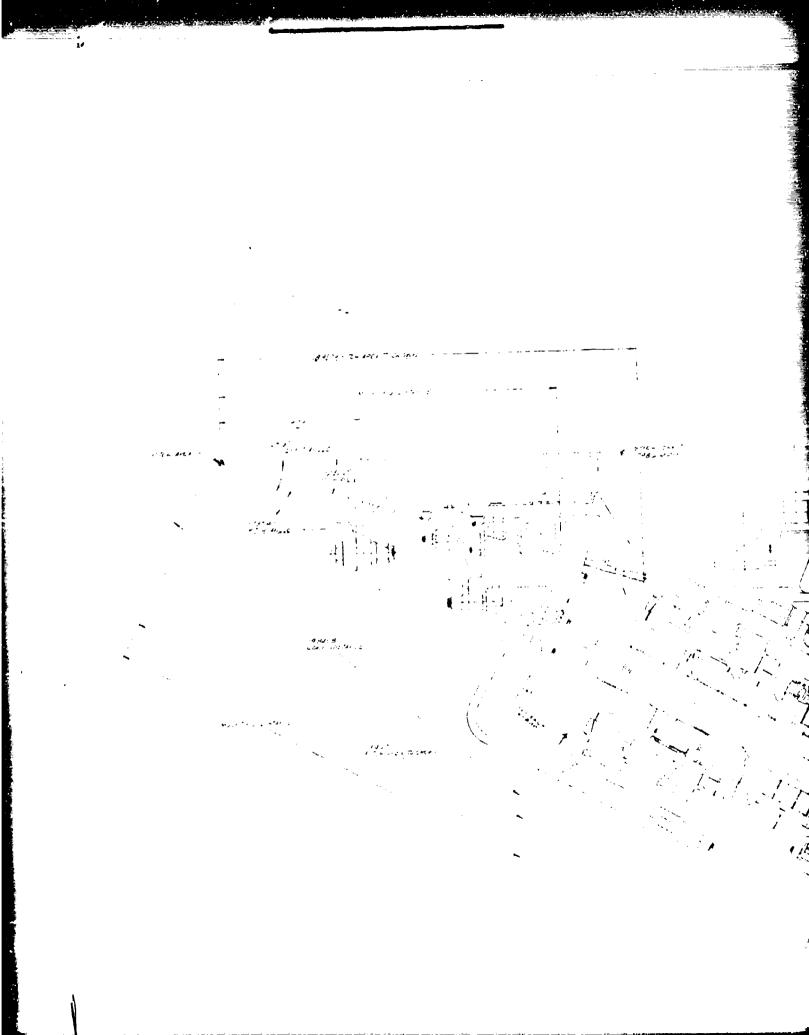
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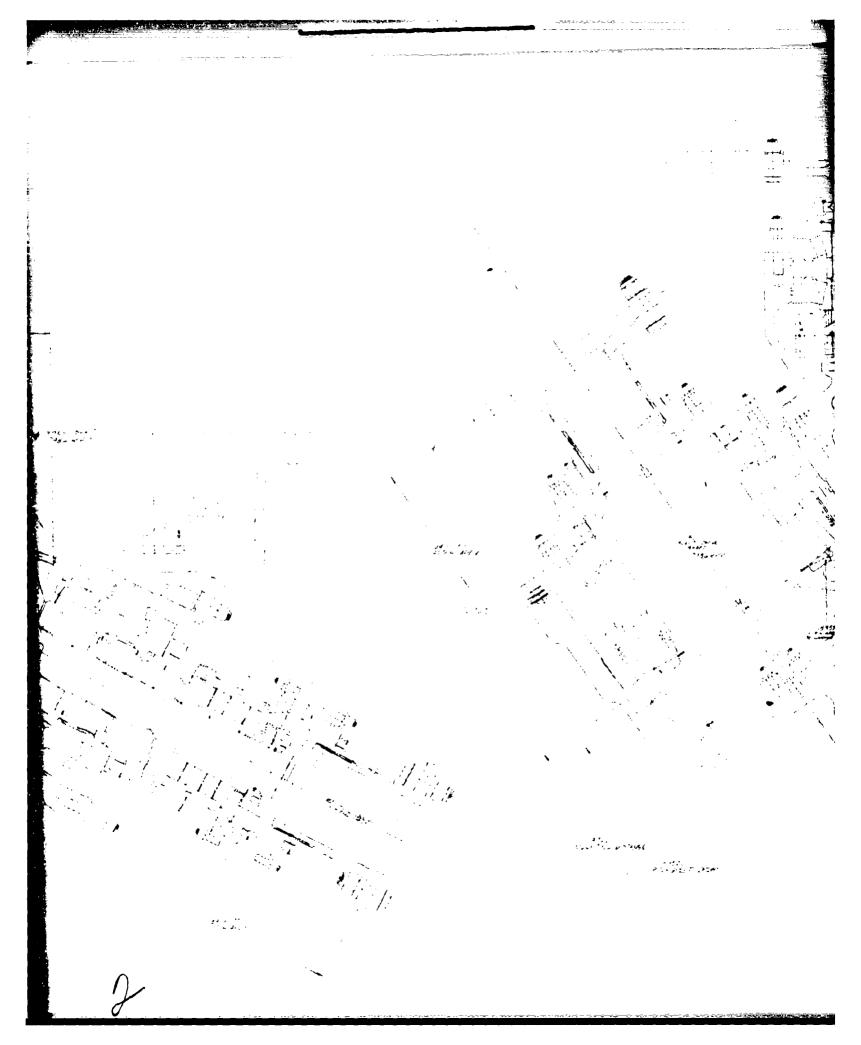
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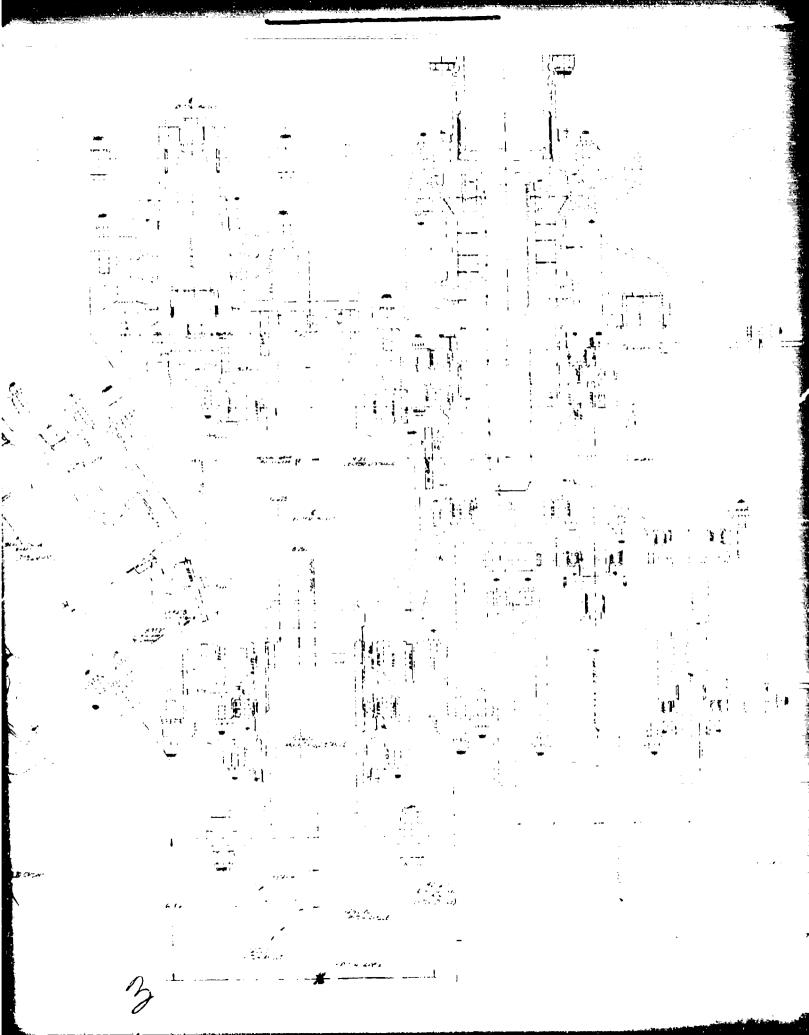
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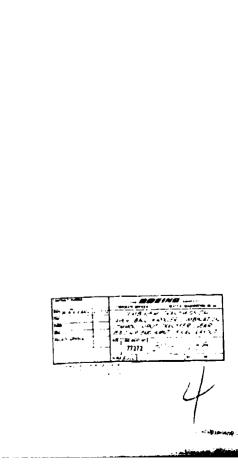
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